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Review

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Zahra Shayegan, Chang-Seo Lee, Fariborz Haghighat

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TiO₂ photocatalyst for removal of volatile organic compounds in gas phase – A review

Zahra Shayegan, Chang-Seo Lee and Fariborz Haghighat*

Department of Building, Civil and Environmental Engineering Concordia University, Montreal, Quebec, H3G 1M8 CANADA

Abstract

Heterogeneous photocatalytic oxidation process (PCO) is a promising technology for removing indoor volatile organic compounds (VOCs) contaminants. Titanium dioxide (TiO2) has been regarded as the most suitable photocatalyst for its cost effectiveness, high stability and great capability to degrade various VOCs. However, no TiO₂-based photocatalysts completely satisfy all practical requirements given photoexcited charge carriers' short lifetime and a wide band gap requiring ultraviolet (UV) radiation. Strategies for improving TiO₂ photocatalyst activities by doping with different metal and/or non-metal ions and by coupling with other semiconductors have been examined and reported. These techniques can improve PCO performance through the following mechanisms: i) by introducing an electron capturing level in the band gap that would generate some defects in the TiO₂ lattice and help capture charge carriers; ii) by slowing down the charge carrier recombination rate and increasing VOCs degradation. This paper reports the outcomes of a comprehensive literature review of TiO₂ modification techniques that include approaches for overcoming the inherent TiO₂ limitations and improving the photocatalytic degradation of VOCs. Accordingly, it focuses on the recent development of modified-TiO₂ used for degrading gas phase pollutants in ambient conditions. Modification techniques, such as metal and non-metal doping, co-doping, and the heterojunction of TiO2 with other semiconductors, are reviewed. A brief introduction on the basics of photocatalysis and the effects of controlling parameters is presented, followed by a discussion about TiO₂ photocatalyst modification for gas phase applications. The reported experimental results obtained with PCO for eliminating VOCs are also compiled and evaluated.

Keywords: Volatile organic compounds (VOCs), Photocatalytic oxidation (PCO), Titanium dioxide (TiO₂), TiO₂ modification, Doping, Composite.

* Corresponding Author: Fariborz.Haghighat@Concordia.ca

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1. Introduction

Volatile organic compounds (VOCs) are a major group of biogenic/anthropogenic indoor air pollutants with several proven adverse health effects [1]. VOCs are not only hazardous compounds, but also participate in undesirable mechanisms with harmful by-products, including organic compounds, ozone, and secondary organic aerosols [2]. To solve this problem, several air purification techniques have been proposed for VOCs removal [3, 4]. Based on the literature, photocatalyst oxidation (PCO) is a promising VOCs removal method [5, 6].

In PCO, VOCs destruction is conducted by using photocatalysts and UV light at ambient temperature [7, 8]. Common photocatalysts (semiconductor materials) used in PCO are TiO₂, ZnO, WO₃, ZnS, and CdS [9, 10]. Up to now, the application of titanium dioxide (TiO₂) for photodegradation of organic contaminates [11-13] as well as batteries and solar cells [14-17] has generated significant attention due to its unique characteristics and environmental friendliness. Additionally, this option is attractive given TiO₂ low cost, simple preparation, high stability, and great capability to degrade various VOCs pollutants [18-20]. Despite its merits, there are some drawbacks, including the large band gap and high charge carrier recombination rate, which restrict its application under visible light or natural solar light irradiation. Given the wide band gap, the electron from TiO₂ valence band can only be activated under ultraviolet (UV) radiation. Furthermore, the quick recombination of photogenerated electrons and holes reduces photonic efficiency and consequently PCO efficiency as well. Therefore, it is imperative to develop highly efficient photocatalysts that separate charge carriers, slow down their recombination rate, and can also be excited under visible irradiation.

Moreover, to enhance the quantum yield of TiO₂ under visible light irradiation, two obstacles must be overcome—the generation of photoexcited charge carriers must be enhanced, and the separation efficiency of these photogenerated charge carriers improved [21-24].

Significant efforts have been made to overcome the aforementioned disadvantages via various strategies such as metal or non-metal doping, co-doping, and coupling with other semiconductors. These techniques seem effective for increasing the lifetime of the photoexcited pairs of charge carriers and for shifting the band gap absorption edges to the visible range of irradiation light [22, 25, 26]. Based on our knowledge, despite considerable reported research on modifying TiO₂-based photocatalysts, there is a lack of a comprehensive review regarding the

effect of modifying TiO₂ by dopants on the removal performance of VOCs in PCO processes in the gas phase. Here, we have provided a comprehensive literature review of TiO₂ modification to present the approaches required for overcoming the inherent TiO₂ limitations and improving the photocatalytic degradation of VOCs. Accordingly, this paper discusses modification techniques. Such techniques include using metal and non-metal dopants, co-doped ions, and composite TiO₂ with other semiconductors for photocatalytic degradation of pollutants in the gas phase. This paper also summarizes the effects of controlling parameters such as relative humidity, air flow rate, UV light, and initial contaminate concentration on VOCs decomposition with TiO₂ photocatalysts.

2. Basic principles of photocatalytic oxidation process

Photocatalyst reactions begin when an electron is stimulated with light energy equal to or greater than the semiconductor's band gap (Fig. 1) [27]. Transferring a photoexcited electron from the valence band (VB) to the conduction band (CB) leaves a hole in the latter, which can be a powerful oxidizing agent. Consequently, electron and hole pairs (e⁻/h⁺) are produced [28-30]. These

pairs can be generated in femtoseconds (fs). Afterwards, photoexcited charge carriers can be trapped within picoseconds (ps) (shallow trap) or nanoseconds (ns) (deep trap) [31]. The hole and electron can recombine in a few tens of nanoseconds [32, 33].

(Figure 1)

Environmental applications of semiconductor photocatalysts are directly related to the interfacial charge transfer mechanism between the semiconductor surface and organic compounds. Furthermore, the mechanism and time scale of interfacial charge transfers and recombination charge carriers could be affected by the interface structure [34, 35]. Moreover, trapped photoexcited holes and electrons exist for nanoseconds and microseconds (µs), respectively [32, 33]. Interfacial charge transfers might oxidize or reduce pollutants directly or produce reactive oxidants like hydroxyl radicals and superoxides [36, 37]. For TiO₂, UV light (about 3% of solar radiation) is a common energy source for PCO because of its wide band gap [38].

The minimum band gap energy required for a photon to generate charge carriers over a TiO_2 semiconductor (anatase phase) is 3.2 eV. This is equivalent to the absorption of a 388 nm wavelength photon. Therefore, TiO_2 photoactivation occurs in $\lambda < 388$ nm, near the UV region [9, 28]. Recently, some researchers have modified TiO_2 to enhance its photoactivity under visible light irradiation [39, 40]. Several studies reported that TiO_2 crystalline form affects its photocatalytic activity [41-43]. TiO_2 has three stable polymorphs—anatase ($E_g = 3.2$ eV), rutile ($E_g = 3.02$ eV), and brookite ($E_g = 3.14$ eV) [10, 44]. For PCO, the anatase performs better than the rutile because its conduction band location drives conjugate reactions involving electrons more efficiently[45]. Combining the rutile and anatase phases has been recommended for increasing TiO_2 photocatalytic activity [27, 43]. Given these facts, Degussa P25 TiO_2 , composed of 75% anatase and 25% rutile, is a commonly-used commercial catalyst [27].

To reduce the recombination rate, it is essential that electron acceptors and photogenerated electrons react [46]. When appropriate surface electron acceptors or a surface defect state are absent, and thus unavailable to trap these charge carriers, electrons and holes pair in a few nanoseconds, reducing the efficiency of photocatalytic degradation. It is reported that the anatase conduction band edge exceeds the rutile by 0.2 eV. This difference facilitates the electron transfer via an electron bridge [47]. As a result, the electron based on the anatase conduction band jumps to the less positive rutile part, thereby reducing the recombination rate of electrons and the formation of positive holes in the anatase site [41, 48].

Using nanocatalysts helps produced electrons reach the surface and react with electron acceptors in a shorter transfer distance. Therefore, the nanostructure recombination rate is lower than the microstructure [46]. Different TiO₂ nanostructure materials such as nanorods, nanofibers, nanospheres, nanosheets, and nanotubes have been produced in the nanostructure, which proves its efficiency [49]. It is proven that various parameters, such as crystallite size, specific surface area, pore structure, pore volume, and crystalline phases, have significant effects on photocatalytic performance [50]. The nano-sized catalyst high surface area and high surface density improves catalyst performance and photoactivity [49, 51].

In general, several factors influence PCO efficiency, such as humidity, flow rate, the initial concentration of contaminant, the characteristics of the photocatalyst, light source, etc. [7, 8]. The effects of above-mentioned parameters are reviewed briefly in Section 4.

3. Issues in VOCs photocatalytic oxidation

For key reactions involved in photocatalytic oxidation, see Table 1. Equation 1 (Photoexcitation) explains TiO_2 activation by UV light [10, 52]. The HOO in Eq. (5) is an electron acceptor that produces hydrogen peroxide (H_2O_2). This generates H_2O_2 as an oxidizing agent and decreases the recombination rate of electrons and holes by capturing photo-induced electrons, thereby enhancing PCO efficiency (Eqs. (6-7)). Reduction and oxidation reactions can both occur on the surface of the photoexcited semiconductor. The main oxidant species for degrading VOCs are hydroxyl radicals produced by oxidizing adsorbed water or adsorbed OH^- . The final results of total mineralization of hydrocarbon compounds by PCO are CO_2 and H_2O [10, 40].

(Table 1)

Photocatalytic oxidation of VOCs includes several processes such as the formation and recombination of electrons and holes, the adsorption of VOCs, chemical degradation, and the desorption of products or intermediates [20]. Moreover, photogenerated electrons and holes may take several pathways. They may recombine on the surface or in the bulk of the semiconductor; electron acceptors may be reduced by electrons; electron donors could be oxidized by holes. Photogenerated electrons and holes could react with O₂ and H₂O to produce reactive oxygen species (ROS) [27]. As mentioned above, most holes are then consumed by reacting directly with adsorbed organic materials or adsorbed water molecules [50]. In some PCO reactions, partial mineralization of parent pollutants can generate some by-products. These intermediates can occupy the catalyst active site and decrease catalyst photoactivity [53].

4. Factors affecting PCO performance

This paper briefly reviews the effects of relative humidity, airflow rate, UV light, and the initial contaminate concentration of VOCs photodegradation. These parameters were also reviewed by Mamaghani [54].

4.1. Relative humidity (RH)

In PCO, the presence of water vapor can be both beneficial and detrimental. Water molecules contribute to photocatalytic oxidation process both as competitive adsorbents and resources for

hydroxyl radicals. The resulting hydroxyl radicals are key players in the oxidation process (Eq. 2). In addition, water vapors can be adsorbed on TiO₂ active sites. This reduces the adsorption of VOCs compounds. Therefore, it is clear that changes in relative humidity (RH) can significantly impact photocatalytic oxidation efficiency [10, 52, 55-59]. Water makes photocatalytic degradation more efficient for some contaminants, like benzene and toluene [56]. In other words, complete mineralization of some pollutants to CO₂ is impossible without water vapor. However, excessive water concentration causes the adsorption of water molecules on the active sites of the catalyst surface and decreases the reaction rate [7, 53, 60, 61]. There are few studies on the effect of relative humidity on doped TiO₂ for gas phase pollutants [39, 62-65]. Earlier studies focused only on the effect of relative humidity on the reaction mechanism of different pollutants. Doped TiO₂ performance with different amounts of dopants in humid conditions has not been thoroughly investigated. Doped TiO₂ could improve photocatalytic efficiency more than undoped TiO₂ in the same humidity. Additionally, the doped TiO₂ catalyst could reduce competition between target compound and water adsorption since the increased separation of electrons and holes produces more oxidizing agents.

4.2. Airflow rate

The airflow rate is a key factor for the photocatalytic oxidation of VOCs. Generally, increasing the airflow rate has two adversary effects. First, reducing residence time during PCO decreases pollutants photodegradation efficiency. Second, increasing the mass transfer rate between the contaminants and catalyst improves the conversion rate [41, 66-68]. Therefore, decreasing the airflow rate increases residence time and gives more VOCs contaminants the chance of being adsorbed on the surface of the catalyst. More adsorbed molecules react with hydroxyl radicals or photo-induced holes and then degrade [7]. According to Han et al. [66] and Yang et al. [67], an increase in flow rate causes a higher mass transfer and a smaller concentration gradient between the bulk and the catalyst surface in laminar regimes. In transitional conditions, the oxidation rate depends on both gas-phase mass transfer rates and surface reactions. It has been reported that the effect of airflow rate on VOCs removal falls under three distinct regimes. [56]. In low airflow rates, increasing the airflow rate enhances VOCs removal. This demonstrates that mass transfer to the catalyst surface limits VOCs elimination. In intermediate airflow rates, flow rate variation has no considerable effect on VOCs removal. This means that surface reaction kinetics is the controlling stage. Finally, in high airflow rates, an increased airflow rate decreases the residence

time for transferring VOCs contaminants from the gas phase to the catalyst surface, thereby decreasing the VOCs removal rate.

4.3. UV light intensity and type

It is worth noting that light intensity is one crucial factor for influencing VOCs reaction rate. Generally, photocatalytic oxidation performance enhances when light intensity increases [10, 69]. Augmenting UV light intensity creates more photons, which provide degradation energy [7]. In general, increasing light intensity augments the production of oxidant species. Moreover, increasing the UV-irradiation intensity heightens photoactivity and consequently enhances the photocatalytic degradation of VOCs pollutants [7, 9, 10, 70]. However, energy loss resulting from light reflection, light transmission, or heat is inevitable in PCO. Attempts have been made to use more energy through different methods such as metal and non-metal ion doping, [9, 71-73], coupling with other semiconductors [60, 74], and surface modification [73]. The recombination of charge carriers, and low interfacial charge transfer rates, are responsible for low quantum yields [29, 75]. The overall quantum yield of light absorbed by photocatalysts or reactants can be estimated using the following equation:

$$\Phi_{\text{overall}} = \frac{\text{rate of reaction}}{\text{rate of absorption of radiation}} \tag{9}$$

where Φ_{overall} is the quantum yield.

Since semiconductors in the heterogeneous medium cannot absorb all refracted irradiation, it is difficult to determine quantum efficiency [9]. Without charge carrier recombination, quantum yield could approach the ideal value of one (1) for UV light with a wavelength small enough to activate the catalyst [76]. VOCs contaminant in the presence of vacuum UV (VUV) lamps can be degraded by photolysis, radical oxidation, and ozonation [77, 78]. Compared with conventional PCO under UV light (254 or 365 nm), VUV light can significantly enhance photocatalytic degradation efficiency. Moreover, by providing a strong oxidation environment and preventing the generation and accumulation of intermediates, VUV light reduces catalyst deactivation [77, 79, 80]. Despite these benefits, performing PCO with VUV lamps produces ozone (O₃) molecules as a by-product. Ozone is a powerful oxidizing species that can react with VOCs pollutants and promote photocatalytic efficiency. However, residual ozone can damage the environment and human health [7, 8, 67, 77, 81].

4.4. Inlet contaminant concentration

The lower initial concentration of contaminants increases removal efficiency in all VOCs. This occurs because the catalyst constant number of active sites has limited adsorption capacity. Furthermore, by-product formation inhibits VOCs adsorption, especially when initial contaminant concentration is high. PCO is therefore more appropriate for photodegrading contaminants in low concentration [7, 82]. The latter can also affect the photocatalyst quantum yield. In fact, when the contaminant is highly concentrated, TiO₂ surface becomes saturated and quantum yield diminishes due to the shielding effect of the contaminant on the catalyst surface [9]. A high concentration of contaminants can also reduce quantum yield by absorbing or scattering light and thus prevent light absorption by the catalyst. Catalyst saturation and competition between adsorbed compounds are higher when inlet contaminants concentration increases [41].

Table 2 provides an overview of TiO₂-based photocatalytic oxidation of various VOCs contaminants and summarizes key operating parameters (e.g. temperature, RH, flow rate, lamp type, and light intensity), maximum conversion, and the main remarks of various studies on VOCs photocatalytic degradation. Accordingly, different types of VOCs, such as alkane, alkene, alcohol, aromatics, ketone, and aldehyde are considered. As Table 2 shows, most studies used UV lamps for all experiments and initial contaminant concentrations were in the range of ppm. One important concern regarding VOCs removal by PCO is the formation of hazardous byproducts. Consequently, it is important to modify TiO₂ for VOCs elimination efficiency. Its modification lessens by-product formation, permits the use of TiO₂ under visible light irradiation and allows the removal of indoors contaminants within the range of ppb.

(Table 2)

4.5. Effect of doping

As stated before, TiO₂ photocatalytic efficiency for environmental applications is limited because of charge carriers high recombination rate. TiO₂ photocatalytic activation is also limited under visible light given its wide band gap [86].

Some promising technologies for modifying TiO₂ that can improve environmental pollutants degradation have been proposed. Techniques include doping TiO₂ with metal or non-metal dopants, co-doping TiO₂ with different elements, and coupling TiO₂ with semiconductors with lower band gap energy. These modifications increase charge separation and shift the light absorption range towards visible light [22]. Generally, an appropriate amount of dopants reduces the band gap and consequently allows light absorption at a higher wavelength [48, 71].

Both metal and non-metal doping methods change TiO₂ electronic structure by inducing new doping levels inside its band gap. The doping levels inside TiO₂ band gap can absorb photons in the visible range, expanding the light absorption range by accepting photogenerated electrons from TiO₂ valence band. This reduces charge carriers recombination rate [23, 87].

See Table 3 for a summary of some preparation methods and a characterization analysis of modified catalysts available in the literature. Among preparation methods reported, the sol-gel technique is the most widely used for its ability to synthesize at ambient temperature and pressure. It also provides good adherence to different substrate shapes [59]. In addition, this process is simple and can control the particle size, shape, properties and product chemical composition [10, 88, 89]. Moreover, it can provide TiO₂ appropriate textural properties, including a highly specific surface area and a porous structure—characteristics that enhance photocatalytic efficiency [88].

(Table 3)

5. Different types of TiO₂ doping

5.1. Metal doping TiO₂

Metal doping is used for introducing additional states in the TiO₂ band gap [92]. With less energy transitions required, these impurity levels cause visible light absorption [107]. TiO₂ band gap impurity levels are induced by substituting metal ions for Ti⁴⁺, closest to the conduction band. As Fig. 1 shows, electrons can be transferred from the TiO₂ conduction band to doping states, or can be excited from TiO₂ valence band to doping states by photons with energy less than 3.2 eV. In addition, metal doping could improve electron trapping by reducing the electron and hole recombination rate [108-110]. The resulting photocatalytic activity and dopant states energy levels strongly depend on several factors, including the dopant chemical nature, its

concentration, and thermal treatment during photocatalyst preparation [107]. It is evident that each cationic dopant presents individual effects on photoactivity. Therefore, the resulting impact may vary depending on the catalyst preparation method or reaction conditions [111]. In general, the preparation method for doped catalysts creates different crystal properties and changes photocatalyst morphology [93, 112]. The different kinds of metal dopants—noble metals, rare earth metals, and transition metals—and their limitations are reviewed in the following sections.

5.1.1. Noble metal and rare earth metal

Noble metals, such as palladium (Pd) [113], platinum (Pt) [91, 94], silver (Ag) [114], ruthenium (Ru) [95], and iridium (Ir) [96], expand TiO₂ light absorption range to visible light. Unlike most base metals, noble metals are highly resistant to oxidation and corrosion in humid air. In the presence of noble metals, charge carriers recombination decreases, which causes photoactivity on the surface of catalyst to increase [91]. Doping noble metal nanoparticles into TiO₂ lattice effectively retards charge carriers recombination as a result of the Schottky barrier formed at the interface of metal-semiconductor (TiO₂) junction. Reducing the electron and hole recombination rate enhances catalysts photoactivity [91, 115]. In fact, the metal nanoparticles form an intermediary for trapping and transferring photogenerated electrons from the TiO₂ surface to electron acceptors [115].

Rare earth metals are a group of 17 chemical elements, including scandium (Sc), yttrium (Y), and 15 lanthanides, in the periodic table. These metals, having incomplete 4f and empty 5d orbitals, could promote catalysis. It has been reported that incorporating rare earth metal ions into TiO₂ lattice could enhance the adsorption of organic pollutants on the catalyst surface, and thus improve TiO₂ photoactivity [116].

Doped TiO₂ photocatalysts with rare earth cations can enhance photocatalysis by increasing light absorption along with modifications of the phase structure, surface area, and surface morphology. Additionally, during the synthesizing process, doping with rare earth ions inhibits the transition from the anatase phase to the rutile phase [117]. Burns et al [26] described that substitutional incorporation of neodymium ion dopants distorts the TiO₂ lattice because Nd³⁺ ionic radius (0.983 Å) is significantly larger than Ti⁴⁺ ion (0.605 Å). Li et al. [118] argued that Nd³⁺ ions could not enter in the TiO₂ lattice and replace Ti⁴⁺ ions; by contrast, Ti⁴⁺ ions replace lanthanide ions in the lanthanide oxides lattice on the Ln³⁺-TiO₂ interface, producing a Ti-O-Ln

bond. Consequently, TiO₂ lattice crystallite size decreases, causing some defects in TiO₂. As a result, finding an optimum amount of dopants is critical for obtaining an appropriate number of defects level.

Recently, doping with metals from the lanthanide group (especially Ce-doped TiO₂) has become an attractive approach. According to Xiao et al. [119], mesoporous Ce-doped TiO₂ has a high surface area, high thermal stability in the anatase phase, and better adsorption capacity for VOCs, which could improve contaminants adsorption on the photocatalyst surface. Korologos et al. [113] studied benzene, toluene, ethylbenzene, and xylene (BTEX) PCO in the gas phase using TiO₂-based photocatalysts. They investigated the effects of different dopants (0.25% (w/w) Pt, Fe, and Ce) on photocatalytic performance. The results illustrated that P25/Ce photocatalyst was the most effective catalyst for benzene and ethylbenzene degradation, and showed comparable performance to the P25 for toluene and m-xylene photocatalytic oxidation.

5.1.2. Transition metal

Researchers have investigated the effects of transition metals such as manganese (Mn) [62], iron (Fe) [73, 86, 90], copper (Cu) [63, 110], vanadium (V) [109], and nickel (Ni) [21] on reducing band gap, decreasing electron and hole recombination rate, and using visible light [10, 90]. The type and amount of transition metals are two decisive parameters for PCO. If the amount of metal dopants remains at an optimum level, dopants can increase the separation of electrons and holes by acting as a charge carrier bridge. However, beyond their optimum value, dopants may distort the catalyst crystallinity and act as a recombination center, reducing PCO efficiency [10, 90]. Therefore, further research is needed to determine the optimal type and concentration of dopant. Fig. 2 compares the influence of different amounts of several metal ions on VOCs conversion, and shows that an optimum concentration of dopants enhances pollutants conversion.

(Figure 2)

Moreover, the photocatalyst surface sites can also be occupied by metal ion dopants, which alter the TiO₂ surface properties depending on dopant type and concentration [63, 109].

Iron. Iron (Fe)-doped TiO₂ is the most popular transition metal-functionalized photocatalyst. Doping Fe into TiO₂ structure decreases electron and hole recombination and reduces band gaps

(Fig. 3). Figure 3 demonstrates that Fe doping narrows the band gap and expands light absorption into the visible range [90]. In addition, iron is much cheaper than noble metals such as Pd, Pt, Ag, Au, and Rh, which explains its widespread use [86]. The main reason for promoting Fe-doped TiO₂ as a catalyst is that the Fe³⁺/Fe²⁺ reduction process enhances charge carrier separation [120]. Fe³⁺ accepts the photoexcited electron, transfers it to the oxygen molecule, and thus produces superoxide [120]. In addition, the radius of Fe^{3+} (0.64 Å) and Ti^{4+} (0.68 Å) is similar, which eases Fe³⁺ ion incorporation into TiO₂ crystal lattice [87, 121]. As a result, Fe ions are distributed uniformly in TiO₂ lattice interstices, which creates defects in TiO₂ structure [121]. The following reactions demonstrated that Fe³⁺ would operate as an electron and hole trap [86, 120]:

$$Fe^{3+} + e^{-} \rightarrow Fe^{2+}$$
 (10)
 $Fe^{2+} + O_2 \rightarrow Fe^{3+} + O_2^{-}$ (11)

$$Fe^{2+} + U_2 \rightarrow Fe^{3+} + U_2 \tag{11}$$

$$Fe^{3+} + h^+ \to Fe^{4+}$$
 (12)

Therefore, the charge carriers recombination decreases and TiO₂ photoactivity enhances.

(Figure 3)

Yang et al. [86] studied iron-doped TiO₂ for BTEX photodegradation on glass fiber (GF). GF-Fe-TiO₂ composites (at different Fe-to-Ti ratios) showed higher photocatalytic activity for BTEX compared to GF-TiO₂ (as a reference) under visible light irradiation [86]. Tieng et al. [90] reported that Fe³⁺ ions could be trapping sites for both photogenerated conduction band electrons and valence band holes. Consequently, the photogenerated charge carriers could be separated more easily and protected for a longer time. As a result, adsorbed species redox reaction occurs more efficiently.

Fig. 2 shows the effect of different amounts of Fe ions in toluene photodegradation [86, 120, 122]. As this figure demonstrates, the optimal amount of Fe ion is around 0.01 wt.%. Increasing the amount of Fe dopant facilitates the transformation of anatase into rutile and increases the rutile phase [121].

Nickel. When Ni is added to the TiO2 structure, it traps photogenerated electrons and separates charge carriers, like an electron acceptor. As a result, Ni²⁺ ions could improve TiO₂ photoactivity [123]. Tseng et al. [64] pointed out that the efficient separation of photogenerated electron and hole pairs depends on the concentration of dopant. In contrast, the dopant may become the

recombination center for photogenerated electrons and holes. In low dopant concentration, doped-Ni²⁺ separates charge carriers by trapping electrons. This increases the likelihood of a hole and VOCs reaction. However, in high dopant concentration, doped-Ni²⁺ acts as recombination center preventing hole formation [64]. According to the following equation, Ni⁺ could transfer electrons to the adsorbed O_2 on the surface of TiO₂ (Eq. 15) and react with Ti⁴⁺ (Eq. 16) to create an interfacial electron transfer [64].

$$TiO_2 + h\vartheta \rightarrow e_{cb}^- + h_{\vartheta b}^+$$
 (13)
 $Ni^{2+} + e_{cb}^- \rightarrow Ni^+$ (14)
 $Ni^+ + O_{2(ads)} \rightarrow Ni^{2+} + O_2^-$ (15)
 $Ni^+ + Ti^{4+} \rightarrow Ni^{2+} + Ti^{3+}$ (16)
 $Ti^{3+} + O_{2(ads)} \rightarrow Ti^{4+} + O_2^-$ (17)

Other transition metals. Huang et al. [79] reported benzene degradation on TiO₂ catalysts with different metal dopants. According to their results, the BET surface area decreased as transition metals (Mn, Co, Cu, Ni, and Fe) were doped in TiO₂ lattice. Transition metals might block TiO₂ pores or cause the particle size to grow, and thus eventually decrease the surface area.

Pham et al. [63] studied benzene photodegradation under visible light on Cu-doped TiO₂, which was supported on porous polyurethane (PU). According to their XPS results, Cu dopants could be substituted by Ti⁴⁺ ions and generate some defect levels on the TiO₂ lattice by forming Cu-O-Ti bonds. Cu-doping TiO₂ improved charge carriers generation. This was possible due to the production of copper oxides (Cu₂O and CuO), which blend with TiO₂ on the photocatalyst surface. Because these copper oxides possess semiconductor properties, they could generate a number of electron and hole pairs under visible light and enhance the photocatalytic degradation of benzene.

Pham et al. [109] investigated toluene degradation on the surface of V-doped TiO₂/PU photocatalysts. The photocatalyst surface in the presence of water vapor could be hydroxylated and form Ti-OH or V-OH bonds. This can happen through the reaction between water and some cations on a photocatalyst surface, including Ti³⁺, Ti⁴⁺, V⁴⁺, and V⁵⁺ ions.

As presented in Table 4, a cation-doped TiO₂ photocatalyst exhibits enhanced photocatalytic degradation performance compared to that of pure TiO₂ or commercial P25 under both UV and visible light. Accordingly, P25 and undoped TiO₂ have no photoactivity under visible light. In

addition, it is obvious that by increasing the small amount of cation dopants, light absorption in the visible range enhances. Furthermore, all cation-doped catalysts exhibited higher photocatalytic efficiency under UV light than under visible light.

(Table 4)

5.2. Non-metal doping TiO₂

Some other researchers have shown that TiO₂ band gap narrowing can be improved by using non-metal (anion) dopants rather than cations [39, 101]. Various studies show that doping TiO₂ with anionic non-metals—nitrogen (N) [19, 24, 39, 71, 106, 120], carbon (C) [38, 97-99], sulfur (S) [65, 101], boron (B) [23], and fluorine (F) [100]—gives promising results. In non-metal doping, dopant states are near the valence band edge and do not act as charge carriers. Therefore, their role as recombination centers might be minimized [126]. Consequently, anion dopants might be more appropriate than cation dopants for expanding TiO₂ light absorption range into the visible region. As Fig. 1 illustrates, impurity levels caused by doping non-metal dopants into TiO₂ lattice are close to the valence band. For an anion-doped TiO₂, the dopants substitute for oxygen (O₂) in the TiO₂ lattice. This expands its light absorption range and narrows the band gap with non-metal dopants p-orbitals (Fig. 1) [10].

Table 5 summarizes a number of anion-doped TiO₂ photocatalysts synthesized to enhance TiO₂ photocatalytic degradation performance. Accordingly, different non-metal elements were considered. Improvements in photocatalytic degradation were observed after adding a small amount of anion dopants. As Table 5 shows, neither undoped TiO₂ or P25 excite under visible light, like in the cation-doping method. Non-metal dopants could change TiO₂ morphology, and increase its photoactivity performance in PCO [71, 72]. In addition, doping with non-metal anions affects the semiconductor electronic structure [25].

Some researchers have reported that thermal treatment during catalyst synthesis caused changes in the catalyst specific surface area and pore volume, which decreased as temperature rose [72, 98]. A number of studies considered the preparation of non-metal-doping TiO₂ by physical or chemical methods, including sol-gel, hydrothermal, hydrolysis, and microemulsion. Table 3

summarizes some of these methods. As this table illustrates, non-metal doped TiO₂ powders are synthesized by applying different preparation methods and precursors.

C- and N-doped TiO₂ nanomaterials have been demonstrated to have superior photocatalytic activity under visible light irradiation compared to other anion dopants [126]. In the following sections C-, N-, and S- doped TiO₂ along with non-metal doping drawbacks are reviewed.

5.2.1. N-doped

The N-doped TiO₂ has received significant attention because it can narrow the TiO₂ band gap and accelerate the charge carrier transfer process [127]. Furthermore, N-doped TiO₂ changes the hardness, refraction index, electrical conductivity, elastic modulus, and photocatalytic activity in the visible light region [10, 71, 106]. Nitrogen has a low ionization potential and an atomic radius comparable to oxygen, meaning N can easily enter the TiO₂ lattice [128]. According to the literature, a general theory suggests that incorporating non-metal dopants with lower electronegativity than oxygen in the TiO₂ lattice shifts 2p-orbital levels of the valence band upward and results in narrower band gap. However, metal doping leads to additional energy states inside the band gap itself (Fig. 4a) [127, 128].

According to Pandian et al. research [128] on electronic band structures for N-doped TiO₂, the highest localized state (acceptor levels) with substitutional nitrogen is 0.14 eV (above the valence band) and 0.73 eV for interstitial nitrogen (Fig. 4a). The substitutional nitrogen is created by mixing N_{2p} orbitals with TiO₂ O_{2p} orbitals. If the latter produces states above the valence band and narrows the band gap, the former correlated interaction between N_{2p} and Ti_{3d} orbitals forms the mid-gap level of N-O states [88, 128]. One problem of N-doping is the significant energy shift in the valence band gap edge occurring in low concentrations of nitrogen. This indicates that, to extend light absorption and improve photocatalytic activity, it is essential to choose the appropriate preparation techniques for N-doped TiO_2 [71]. Albrbar et al. [88] reported that nitrogen can be incorporated in the TiO_2 lattice structure as a N^{3-} anion interstitially, or by substituting O^{2-} anions [88]. According to Zeng et al. [19] XPS results for N-TiO₂, three peaks are identified for N_{1s} : 396.1 eV, 400 eV, and 403.6 eV. The first N_{1s} peak (396.1 eV) is attributed to the substitutional nitrogen dopant since the 395.6 eV is characteristic of N^{3-} corresponding to the Ti-N bond. The second peak (400 eV) at high binding energies is

attributed to interstitial nitrogen dopants, and the third one (403.6 eV) is usually defined as NOx species. This indicates the successful incorporation of nitrogen into the TiO₂ lattice [19].

5.2.2. C-doped

C-doped TiO₂ can significantly stabilize anatase TiO₂ and enhance the adsorption of organic contaminants on the catalyst surface. It also increases TiO₂ conductivity, which accelerates the charge transfer from inside TiO₂ structure to the surface to participate in oxidation reactions [98, 99]. Di Valentin et al. [129] studied how replacing oxygen sites with carbon, or putting carbon into the interstitial sites of TiO₂ lattice, enhances visible light absorption. There is a debate around carbon-doped TiO₂ similar to that around nitrogen doping on whether the doped type of carbon is substitutional or interstitial. Furthermore, the band gap can also be narrowed with residual carbon-containing species. These have a complex composition given the partial removal of organic compounds in the calcination procedure [126].

As Fig. 4b demonstrates and according to Khalilzade et al. [130], C-doped TiO₂ mid-gap band position shifts upward when carbon concentration increases, which shows a high oxidative potential for photogenerated holes. The new electronic level above TiO₂ valence band was responsible for narrowing C-doped TiO₂ band gap and visible light photoactivity. Wang et al. [99] studied C-doped TiO₂ at different calcination temperatures. They pointed out that C-doping added the deepest states to the band gap compared with N and S-doping (Fig. 4).

Choosing an appropriate method for catalyst preparation is critical for reducing the number of TiO₂ lattice defects. The latter act as great recombination centers for charge carriers and limit the visible light photocatalytic efficiency [98]. Different C-doped TiO₂ preparation methods can be seen in Table 3.

5.2.3. S-doped

S-doped TiO₂ has been investigated extensively in environmental (air and water pollution) uses [101, 131]. Introducing S into the TiO₂ lattice narrows the band gap. Mixing S_{3p} states with O_{2p} of TiO₂ valence band was demonstrated to help increase the valence band width, and narrow the band gap too (Fig. 4c) [131]. S dopant ions can be incorporated into the TiO₂ lattice either as a S^{2-} anion. In S-doping, S^{2-} is replaced with lattice oxygen (O₂), or as a cation, where S^{4+}/S^{6+} substitutes Ti^{4+} . [88, 132]. Ma et al. [131] reported that Ti^{4+} ion atomic radius (0.64 Å) is relatively greater than S^{6+} ion (0.29 Å). Therefore, in S-doped TiO₂ catalyst when S^{6+} substitutes

 Ti^{4+} might decrease the catalyst crystal size. Jo et al. [65] reported that TEM images for S-doped TiO_2 show less agglomeration than those of TiO_2 Degussa (un-doped). Furthermore, SEM images of both catalysts illustrate that S-doped TiO_2 crystalline particle was smaller than undoped TiO_2 or Degussa P25 [65]. According to Yu et al. [133], doping TiO_2 with anionic S could be more difficult than with cationic ones given the different ionic radius of S^{2-} (1.7 Å), which is significantly larger than that of O^{2-} (1.22 Å). Consequently, Ti-S bonding requires larger formation energy than Ti-O bonding. Ti^{4+} is therefore replaced with S^{6+} as this union has chemical advantages over O^{2-} substitution by S^{2-} . As Fig. 4c explains, by substituting Ti^{4+} with S^{6+} , S_{3s} orbitals create states above the O_{2p} orbitals of TiO_2 valence band, and S_{3p} orbitals contribute to lowering TiO_2 conduction [131]. The ionic form of sulfur dopant depends on the preparation routes and sulfur source. For instance, the anionic form of S dopant was produced when TiS_2 was used as the source. In that case, S in TiS_2 was oxidized and the residual sulfur remains as S^{2-} [133].

(Figure 4)

(Table 5)

5.3. Co-doping TiO₂

Some researchers have focused on co-doping as one way to compensate for the drawbacks of single-doped TiO₂. This approach is explained in detail in the following sections. Various studies have performed TiO₂ co-doping by combining metal/non-metal and non-metal/non-metal dopants discussed below. However, few studies have considered metal/metal TiO₂ co-doping [108].

5.3.1. Metals co-doped

The physiochemical properties and doping mechanism of combining two different metals to codope TiO_2 have not been thoroughly investigated. Pham et al. [108] studied hexane and butyl acetate photodegradation on Ag/V co-doped TiO_2 deposited on polyurethane. According to their report, Ag^+ ions radius (1.26 Å) is much bigger than Ti^{4+} ions. In the case of V-doped TiO_2 , due to the similar radii of V^{4+} (0.72 Å) and Ti^{4+} (0.74 Å) ions, vanadium was incorporated more easily into the TiO_2 lattice. V^{4+} ions could therefore substitute Ti^{4+} ions to form Ti-O-V bonds in

the TiO₂ lattice. Ag and V were incorporated into the co-doped TiO₂, which formed Ti³⁺. Consequently, the Ti³⁺/Ti⁴⁺ ratio of the Ag-V co-doped TiO₂ was higher than that of a single V or Ag-doped photocatalyst. This indicates that compared to single doping, Ag-V co-doped TiO₂ enhanced dopants incorporation, therefore increasing oxygen vacancies (or Ti³⁺ formation).

In other words, incorporating metal dopants produced oxygen vacancies in the TiO₂ lattice, reducing Ti⁴⁺ into Ti³⁺. Moreover, the Ag and V co-doping TiO₂ could prevent the aggregation of Ag, Ag₂O, and V₂O₅, creating uniformly distributed particles on the surface of TiO₂ in the Ag-V co-doped TiO₂/PU photocatalysts. Additionally, in Ag and V co-doping TiO₂, vanadium increased the internal electron transfer in the TiO₂ lattice and silver enhanced internal electron transfer in the Ag₂O. Vanadium also improved exterior electron transfer between Ag₂O and TiO₂ [108]. Zhang et al. [134] reported that adding alkali metals ions (Li⁴, Na⁺, and K⁺) to Pt/TiO₂ catalysts caused more atomically dispersed Pt species; improved the H₂O and O₂ activation efficiently; and promoted catalysts performance in HCHO oxidation. In another study on HCHO oxidation, Zhang et al. [135] found that Na addition also had a visible promoting effect on Pd/TiO₂ catalysts due to the good dispersion of Pd species.

5.3.2. Non-metals co-doped

Non-metals co-doped TiO₂ has been studied extensively [72, 120, 136]. The S-N co-doped TiO₂ exhibited strong absorption ability in both UV and visible light regions [10]. According to the XPS spectra of N, S and C tri-doped TiO₂ (NSC-TiO₂) in Lei et al. [136] research, N was assigned to the Ti-O-N and Ti-N bonds; S was mostly attributed to the Ti-O-S bond; and C was ascribed to the Ti-O-C bonds in the NSC-TiO₂ catalyst. In addition, results showed that co-doping and calcination temperature affects the catalyst microstructure and crystallite size. Li et al. [137] studied the degradation of gaseous acetaldehyde, toluene, and trichloroethylene in N and F-co-doped TiO₂ photocatalytic process. They found that co-doping nitrogen and fluorine has a synergetic effect. N doping enhanced visible light absorption, photocatalytic activity, and the formation of superoxide (O₂•-) radicals through the oxygen vacancies sites. F-doping formed new active sites and hydroxyl radicals (OH•).

Non-metals can be introduced into the TiO₂ lattice by applying various precursors and preparation methods (Table 3). In C and S co-doped or S-doped TiO₂, thiourea is generally used

as a non-metal doping source [72]. In the initial thermal treatment of catalyst preparation, thiourea was first decomposed by increasing the temperature. This temperature increase released N-containing gas, which can interact with Ti and O atoms to produce N dopants into the TiO_2 lattice. During the thermal treatment, many oxygen vacancies are produced and N-containing gas can easily enter the vacancies [72]. The formation of oxygen vacancies in N-doping TiO_2 happens because two nitrogen atoms should replace with three oxygen atoms to maintain electro neutrality, and so produce an oxygen vacancy in the TiO_2 crystal lattice. Furthermore, superoxide radicals $(O_2^{\bullet-})$ that require oxygen vacancy sites form [137].

5.3.3. Metal and non-metal co-doped

Dopant components in the TiO₂ lattice, including metal and non-metal, produce doping energy levels in the TiO₂ band gap that improve visible light absorption and result in higher photocatalytic performance (Fig. 5). Furthermore, metal components prefer to substitute for the Ti site in the TiO₂ lattice to create the dopant level near the conduction band. Non-metal components can potentially form new levels closest to the valence band that reduce the band gap and cause visible light absorption. Therefore, both metal and non-metal ion co-doping enhance photocatalytic activity [120, 138]. The synergistic effects between metal and non-metal dopants improve electron and hole separation, enhancing TiO₂ photocatalytic activity under visible light [120, 138].

(Figure 5)

It has also been reported that TiO₂ photoactivity when doped with nitrogen or sulfur can be improved by adding Fe(III) as a co-dopant [23]. N-Fe co-doped TiO₂ was studied by Dong et al. [120] in order to modify N-TiO₂ surface by Fe-ions. The results illustrated that a small amount of Fe-ions (0.02 wt.%) added to N-TiO₂ nanocrystal improved the catalyst photoactivity under visible and UV irradiation. It also improved augmented photodegradation efficiency to 97%. In addition, Fe-ions modification can also improve N-doped TiO₂ stability. In other words, N-doped photocatalyst instability by nitrogen loss happens when the photogenerated hole oxidizes the catalyst lattice nitrogen and gradually leads to deactivation. However, the lattice nitrogen was relatively stable due to Fe-ions surface modification in Fe/N-doped TiO₂ [120]. Li et al. [138] studied the C, Mo co-doped TiO₂ and reported that after Mo-doping TiO₂, the doping energy level was 0.4 eV. This level was extremely positive compared to the TiO₂ conduction band.

Consequently, the valence band electrons could be excited to the Mo⁶⁺/Mo⁵⁺ doping energy level under visible irradiation. Li et al. [138] also argued that the recombination rate in C, Mo codoped TiO₂ is lower than that of Mo-doped TiO₂.

Table 6 summarizes a number of co-doped TiO₂ photocatalysts synthesized to improve TiO₂ photocatalytic degradation performance. Different types of co-doped TiO₂ catalysts, including non-metal/non-metal, metal/metal, and metal/non-metal performance in the PCO of gas phase pollution and their operating parameters, are reviewed in Table 6.

(Table 6)

5.4. Composite semiconductors

One effective technique for improving TiO₂ photoactivity is coupling it with other semiconductors that generally have lower band gap energy. As a result, TiO₂ photoactivation expands to the visible region. Coupling catalysts are applied to accelerate electron and hole separation [36]. These semiconductors include WO₃ [106], SiO₂ [37], SnO₂ [21, 22, 55], MnCO₃ [139], CdS [140], Ag₃VO₄ [36], ZnO [141], and porous material with a large surface area (e.g. activated carbon) [10]. As illustrated in Table 7, semiconductors coupled with TiO₂ performed an enhanced photocatalytic degradation compared pure TiO₂. Different metal/non-metal doped with semiconductor composites are considered for improving TiO₂ photoactivity.

Among the different semiconductors reviewed, SnO₂ is interesting for having lattice parameters and a tetragonal structure similar to TiO₂. As Fig. 6a shows [21, 22, 142], in the TiO₂ and SnO₂ heterojunction, TiO₂ photogenerated electrons move to SnO₂ conduction band while holes transfer in the reverse direction, thereby separating charge carriers and suppressing recombination as a result [21, 22]. Substituting Sn (SnO₂) for Ti (TiO₂ lattice) can only cause a blue shift in TiO₂ light absorption spectra since SnO₂ band gap is larger than TiO₂. Consequently, to obtain visible light photoactivity, there should be a surface coupling between the two semiconductors [21].

Liu et al. [140] studied the degradation of benzene on CdS-TiO₂, which was supported on fiber glass cloth. TiO₂ and CdS can be excited under UV-vis irradiation to produce electron and hole pairs (Eq. 18). According to the electronic structure of mentioned semiconductor, the TiO₂

conduction band is more positive than that of CdS nanoparticles, which leads to the formation of a local electric field (Fig. 6b) [140, 142]. As a result, photogenerated electrons could transfer from CdS nanoparticles conduction band to the TiO₂, while the photogenerated holes in the TiO₂ semiconductors valence band accumulate in the CdS valence band to create a hole center (Eq. 19). Consequently, the recombination rate of charge carriers can be reduced effectively. For visible light illumination, only CdS nanoparticles can be excited and photogenerated electrons moved to the TiO₂ conduction band (Eqs. 20 and 21) [140].

$$CdS - TiO_2 \xrightarrow{UV - vis \ light} CdS(e^- + h^+) - TiO_2(e^- + h^+)$$

$$\tag{18}$$

$$CdS(e^{-} + h^{+}) - TiO_{2}(e^{-} + h^{+}) \rightarrow CdS(h^{+}) - TiO_{2}(e^{-})$$
 (19)

$$CdS - TiO_2 \xrightarrow{Visible \ light} CdS(e^- + h^+) - TiO_2$$
 (20)

$$CdS(e^{-} + h^{+}) - TiO_{2} \rightarrow CdS(h^{+}) - TiO_{2}(e^{-})$$
 (21)

Some studies have also focused on TiO₂/SiO₂ composite [37, 143-145]. Yu et al. [143] argued that in TiO₂/SiO₂ composite thin films, increasing SiO₂ percentages augments the surface hydroxyl. Consequently, the growth in surface hydroxyl content will trap more holes in the valence band and prevent charge carriers from recombining. In addition, TiO₂/SiO₂ photoactivity increased when SiO₂ was less than 5 mol% and decreased when higher than 10 mol%. Zou et al. [145] claimed that TiO₂/SiO₂ composite pellet shape had a higher adsorption capacity than thin film. Consequently, they acted as both photocatalyst and adsorbent. Zhan et al. [37] reported that in TiO₂/SiO₂ composites, increasing the doped silica molar ratio enhances the surface area of composite fibers. The larger surface area increases photocatalytic degradation efficiency. By contrast, the Ti–O–Si bond came out and resulted in lots of chemical defects on the surface of SiO₂ doped TiO₂ fibers, which can also enhance the oxidation rate. However, an excessive amount of high silica ratio recombines charge carriers and decreases the degradation rate [37].

Lee et al. [106] investigated the N-doped TiO₂/WO₃ composites photocatalytic degradation efficiency. The catalyst photoactivity enhanced with an increase in WO₃ loading from 0.5% to 4.0%. However, an additional increase in WO₃ loading to 6.0% reduced photocatalytic efficiency. Fig. 6c shows the electronic structure of composite WO₃-TiO₂, and demonstrates that electron and hole pairs could move between these semiconductors bands [106, 142].

As seen in Fig. 6d [146] and according to Ferrari-Lima et al. [105], the heterojunction of TiO₂ and ZnO semiconductors (which have similar a band gap), creates synergic effects, decreases the recombination rate, and increases charge carriers lifetime. For TiO₂-ZnO, the electron transfers

from ZnO conduction band to that of TiO₂, and the hole transfers from TiO₂ valence band to that of ZnO. As a result, the recombination rate of electron–hole pair decreases, and the charge carrier lifetime increases [105, 146, 147].

(Figure 6)

(Table 7)

6. Challenges

6.1. Limitation of doping

6.1.1. Limitations of metal doping

As pointed out earlier, metal doping TiO₂ impacts photocatalysis performance positively. However, earlier research has shown that doping TiO₂ with metal ions may have considerable drawbacks. Transition metals might cause partial blockage on TiO₂ porous surface sites [119] or cause the growth of particles. This would decrease the specific surface area and PCO performance subsequently [79]. As Table 4 and Yang et al. research [86] show, BTEX degradations on Fe-doped TiO₂ decreased under UV light because of a probable blockage, and therefore decreased in surface area. In addition, Korologos et al. [113] reported that benzene, toluene, and m-xylene degradation was not efficient in Fe/P25 and Pt/P25 under UV light irradiation.

Additionally, metal-doped TiO₂ catalysts have shown thermal instability [127, 128] and photocorrosion [39]. Metal dopants may act as recombination centers, encouraging the recombination of photogenerated charge carriers [39, 87, 127]. When metal loading concentration is beyond optimum value, electron transfer from TiO₂ to metal dopants could deform the potential field in TiO₂ particles and charge metal centers negatively. This would draw a portion of holes near the metal—TiO₂ junction (especially for highly doped catalysts), and heighten the recombination rate. The deposited metal could also shield from photon absorption, which is detrimental to photocatalytic activity [148]. Sun et al. [73] reported that, for Fe-doped TiO₂, the separation and transportation efficiency of charge carriers enhanced when Fe³⁺ doping was increased from 0.1% to 0.7%, and significantly decreased when increased from 0.7% to 1.5%.

Moreover, although noble metals and rare earth metals could efficiently improve PCO, the cost-effectiveness required for industrial uses usually leads to more economical transitions or non-metals dopants being chosen.

6.1.2. Limitation of non-metal doping

Despite non-metal-doped TiO₂ photocatalytic performance with visible light, the photocatalyst presents some disadvantages. Doping non-metal dopants into the TiO₂ lattice causes oxygen vacancies to form in the bulk, which may become an important recombination center for charge carries. These defects limit visible light photocatalysis efficiency significantly [98]. The catalyst preparation method is one key parameter that should be considered for photocatalytic uses. Additionally, the doping procedure of non-metal dopants requires thermal treatment at high temperatures (400-850 °C) or a long preparation time requiring high energy (see Table 3). Moreover, using toxic, expensive, or unstable precursors and the formation of undesirable gaseous by-products during the synthesizing process makes the preparation process costly and hinders large-scale production [38, 98]. Furthermore, the content of non-metal dopants would decrease during the annealing process and reduce visible-light photoactivity [149].

6.2. Stabilization by support structure

Catalyst immobilization is one key factor for heterogeneous PCO efficient removal of indoor air pollutants. Applying an effective coating method and substrate type, two critical parameters in catalyst immobilization, is important. Appropriate substrates should have stable physical and chemical properties in photochemical oxidation reactions. In addition, the substrate should provide a large surface area for the catalyst to coat on. Other essential properties of the substrate are: sufficient adhesion of the TiO₂ particles; resistance against sintering temperatures; and a strong absorbance affinity with contaminants. Different materials, such as activated carbon fibers [84], glass [65, 113], various polymeric materials [63, 108, 109], zeolites [150], fiberglass fiber [61, 81], carbon cloth fiber [61, 81], stainless steel [151], carbon nanotube [152], carbonized cotton fibers [59], and glass fibers (GFs) [86] have been used as support.

Glass and silica substrates are among the widest used materials because they are transparent. Activated carbon has high porosity, which creates a high specific surface area and enhances photocatalyst activity. Carbonized cotton fibers (CCFs) are an alternative support for TiO₂. CCFs

have great corrosion and temperature resistance, high conductivity, and strength. In addition, glass fibers (GFs) are flexible and easier to handle than other supports [59, 86].

Pham et al. [109] considered gaseous toluene degradation on V-doped TiO₂. Porous polyurethane (PU) was used as a support for immobilizing the photocatalyst and so increases its adsorption ability to remove toluene under visible light irradiation. In another study [63], they used PU as a substrate for Cu doped TiO₂ for benzene degradation. Immobilized on a porous support, Cu-TiO₂ showed improvement in both adsorption capacity and photocatalytic oxidation for benzene removal. Jo et al. [101] studied the aluminum sheet-based S-doped TiO₂ for BTEX photodegradation. Aluminum is considered a flexible low-cost substrate. Yang et al. [86] stated that stabilizing the nanocatalyst on the substrate in air purification system is essential. They considered BTEX photodegradation on Fe-doped TiO₂, which supported on the glass fiber. They examined the applicability of glass fiber supported photocatalyst for eliminating VOCs under visible and UV light irradiation. Zhong et al. [7] studied the photocatalytic oxidation process of TiO₂ coated on fiberglass fibers (FGFs) and carbon cloth fibers (CCFs). In this study, the photolysis effects under vacuum ultraviolet (VUV) and ultraviolet (UVC) irradiation for a different kind of VOCs were investigated in a pilot duct system. The single-pass removal efficiency of considered compounds showed the following sequence: TiO₂/CCFs+VUV > TiO₂/FGFs+VUV > TiO₂/FGFs+UVC. According to their results, coating nano-TiO₂ particles on the CCFs provided a large surface area that produced high photocatalytic activity. In a different study, Zhong et al. [61] reported that TiO₂ /FGFs catalyst possesses hydrophilic characteristics. Because of this, the adsorption coefficient for polar VOCs was higher than that of non-polar VOCs at the same RH level. Moreover, the BET surface area of TiO₂/CCFs being higher than that of TiO₂/FGFs led to a greater adsorption capacity [61].

A desirable coating technique should have the following requirements: a) a durable and stable coating; b) an efficient and appropriate contact between the catalyst and the contaminant; c) be non-selective for different substrates; and d) be cost-effective and suitable for large-scale application [153]. Generally, the dip coating and spray coating are used for immobilizing TiO₂ on the catalyst support [6, 40, 154]. Dip coating forms a thin and controllable layer, while spray coating may be used to create a thicker layer [59].

6.3. Application in industrial, civil and building construction

General studies considering the PCO for removal of a wide range of VOCs in the gas phase are reviewed in this paper. However, most reported studies are based on bench-scale operating in laboratory conditions and there are only a few papers dealing with full-scale or industrial application of PCO technology [7, 53, 61, 155, 156]. Although, the acceptable efficiency of PCO is proved in laboratory scales, there are some limitations during the scaling up the system which are operating in real conditions [157]. For example, the most reported research were done for a single compound with low airflow rate (in L/min level) and high VOCs concentration (in hundreds ppm level), which is far from real indoor applications [158]. Consequently, results from these research may not appropriately accurate and practical for the design of large-scale application of PCO for removal of VOCs [158].

Despite enormous studies on UV-PCO for removal of VOCs in the laboratory, limited research has been studied the catalyst deactivation problem, by-product production and moderate removal efficiency [158]. By-products and intermediates production and their accumulation on the surface of catalyst may cause the deactivation of catalyst [156, 159]. An additional concern that restricts full-scale applications of PCO in air purification system is their moderate efficiency, particularly when they are operating in high flow rate that is more similar to the real application [158]. Improvement of the catalyst surface characteristics in order to reduce the catalyst deactivation may address this limitation. The deactivation may be reduced through deferent strategies: a) expand the light absorption edge into the visible range that more photon could be used; b) reduce the recombination rate of charge carriers to enhance the quantum yield; c) design the PCO reactor with the configuration that the maximum light reach to the full surface of catalyst [158].

The large-scale applications of photocatalytic process were used for air purification and self-cleaning purposes for the first time in Japan, in middle of the 1990s [157]. One of the most successful self-cleaning coatings, which are used in the practical applications, is TiO₂-based photocatalytic coating. Many substrates such as window glass for building, concrete for construction and bridges, metal for traffic infrastructure and architecture, and polymers for

outdoor objects have been modified with TiO₂-based coating to achieve self-cleaning performance [160].

Zhong et al. [7] studied different VOCs removal in a pilot-scale application. They considered the operational parameters such as contaminant concentration, relative humidity, and residence time quite similar to the real conditions. They also reported the generated by-products for each compound. He et al. [156] studied the paint plant's waste gas removal in a pilot-scale system combined with PCO and biotrickling filtration (BTF). They reported that the combination of PCO and BTF is efficient for the removal of the high concentration and multicomponent VOCs from waste gas of paint plant.

7. Conclusion

PCO is a promising technology for air purification that can eliminate indoor air contaminants effectively at room temperature. PCO degrades hydrocarbon contaminants to carbon dioxide and water. However, in most cases, some hazardous by-products will inevitably form during PCO. Various factors such as humidity, light type and intensity, photocatalyst concentration, contaminant concentration, and flow rate can affect PCO efficiency.

TiO₂, the most used semiconductor photocatalysts in PCO, requires UV light irradiation to activate given its relatively wide band gap. As a result, visible light does not have sufficient energy to excite the electrons from the valence band to the TiO₂ conduction band. Doping TiO₂ with metal, non-metal and different composite semiconductors are promising methods for overcoming its inherent limitations and for enhancing photocatalytic activity under visible light irradiation. In addition, the effective reduction of electron and hole recombination rate is another result of doping.

Metal components create the dopant state near the conduction band, however, non-metal components form the new state closest to the valence band that lower the band gap and expand the TiO₂ light absorption edge. Moreover, the synergistic effects between metal and non-metal dopants in co-doped TiO₂ improve electron and hole separation, followed by enhancing TiO₂ photocatalytic activity under visible light. Coupling TiO₂ with other semiconductors that generally have lower band gap energy improves the TiO₂ photoactivity, accelerate electron and hole separation, and expands its light absorption to the visible region.

Furthermore, it is essential to find an optimum amount of dopant to increase the separation of charge carriers and prevent the formation of a recombination center. Additionally, choosing an appropriate synthesizing or coating method to improve the catalyst features should also be considered.

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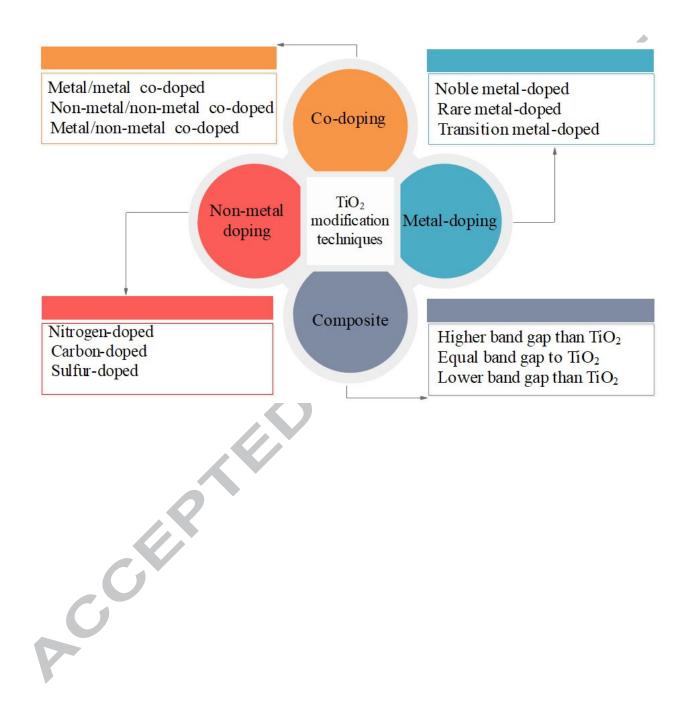
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Highlights

- The limitations of the TiO₂-based photocatalyst for photocatalytic degradation of VOCs are reviewed.
- The approaches for modifying TiO₂-based photocatalyst to improve the performance of photodegradation of VOCs are discussed.
- Applications of TiO₂ and modified-TiO₂ photocatalysts for VOCs removal in the gas phase are summarized.
- Fundamental of metal and non-metal doped TiO₂, co-doped TiO₂, and composite TiO₂ with other semiconductors are considered.
- The effects of key controlling parameters on PCO efficiency are reviewed.



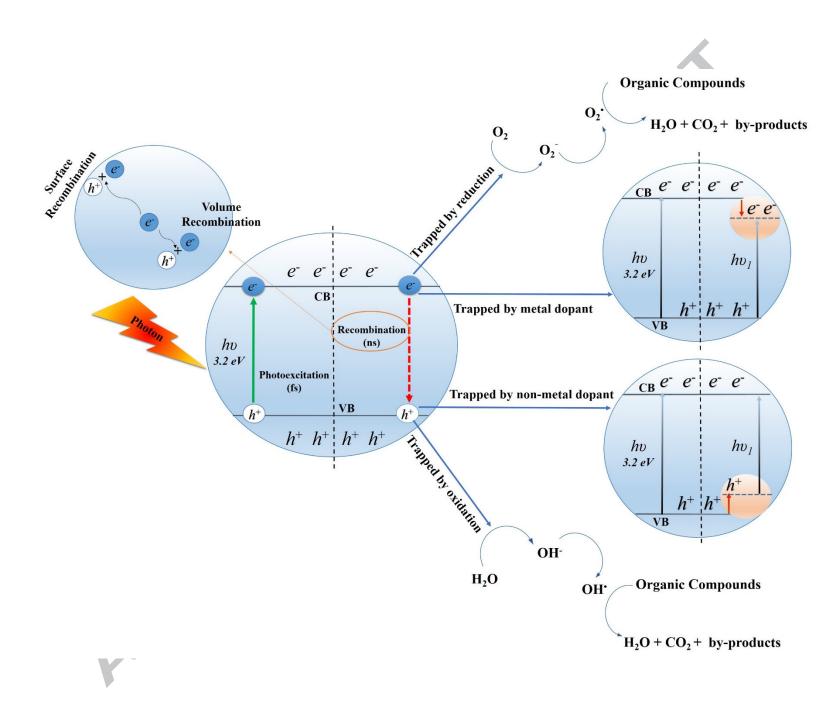


Figure 1. Basic principle of PCO for removal of VOCs.

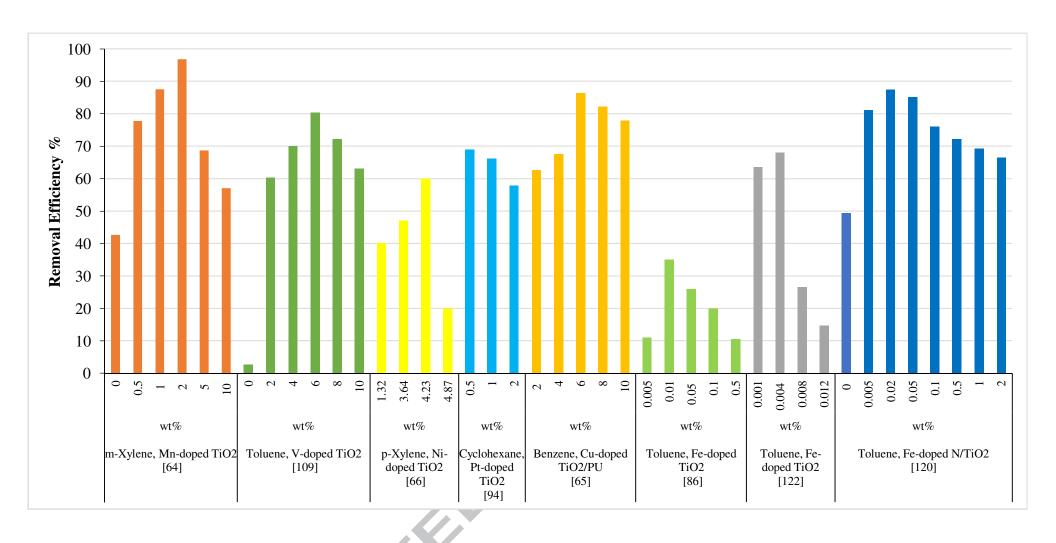


Figure 2. Influence of the amount of different metal dopants (wt.%) on VOCs degradation by TiO₂ photocatalyst.

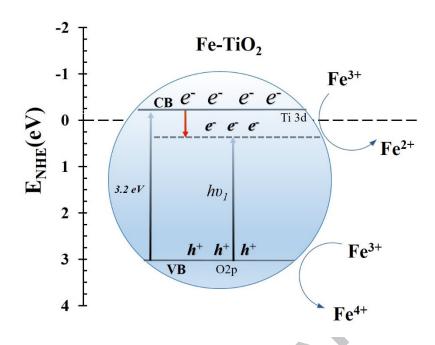


Figure 3. Schematic energy level of Fe (transition metal) doping TiO_2 .

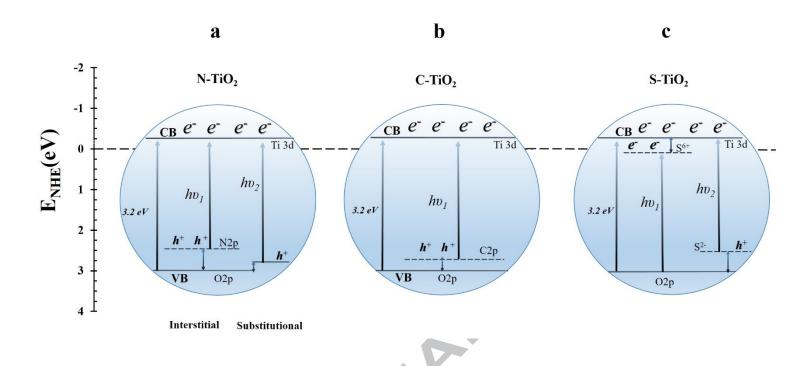


Figure 4. Schematic energy level of (a) N-doped TiO_2 , (b) C-doped TiO_2 , (c) S-doped TiO_2 .

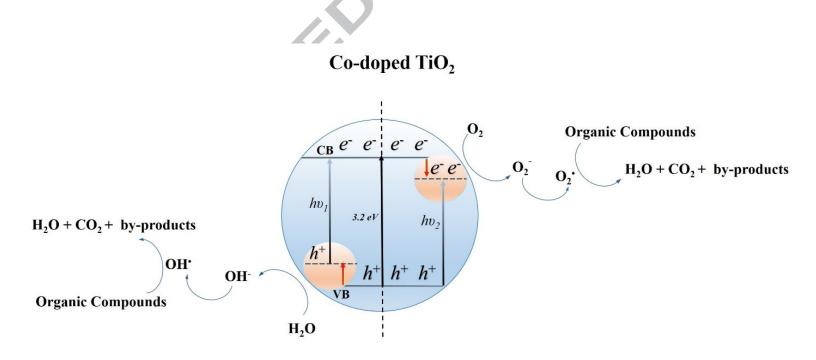


Figure 5. Schematic energy level of metal and non-metal doped TiO₂.

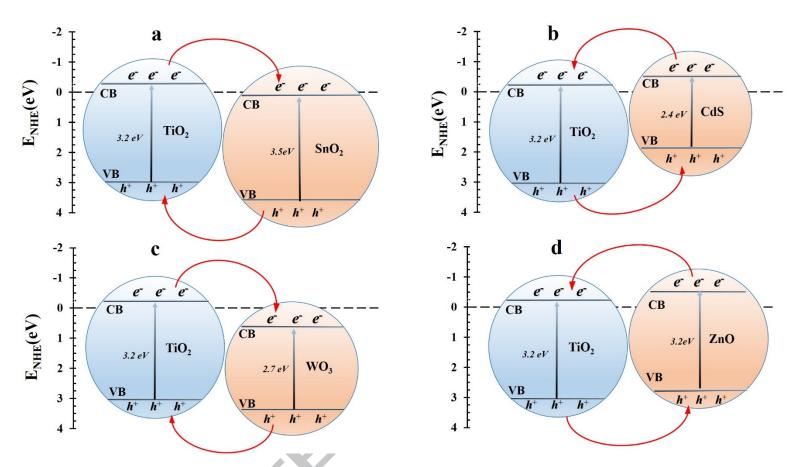


Figure 6. Figure 6. Schematic energy level of TiO₂ composite (a) SnO₂ (data extracted from [21, 22, 142]), (b) CdS (data extracted from [140, 142]), (c) WO₃ (data extracted from [106, 142]), (d) ZnO (data extracted from [146]).

Table 1. The reactions involved in the TiO_2 photocatalytic oxidation process for the degradation of VOC pollutants.

Reactions	Eq. No.
Photoexcitation: $TiO_2 + h\vartheta \rightarrow h^+ + e^-$	1
Oxidation reaction: $OH^- + h^+ \rightarrow OH^{\bullet}$	2
Reduction reaction: $O_{2(ads)} + e^- \rightarrow O_{2(ads)}^-$	3
Ionization of water: $H_2O \rightarrow OH^- + H^+$	4
Protonation of superoxide: $O_2^{\bullet-} + H^+ \rightarrow HOO^{\bullet}$	5
electron scavenger: $H00^{\bullet} + e^{-} \rightarrow H00^{-}$	6
Formation of H_2O_2 : $HOO^- + H^+ \rightarrow H_2O_2$	7
$OH^{\bullet} + pollutant + O_2^- \rightarrow products(CO_2, H_2O, etc.)$	8

^{*}where h^+ and e^- are hole and electron, respectively.



Table 2. Overview of photocatalytic degradation of VOCs over TiO_2 in various conditions.

Contaminant (concentration)	Photocatalyst	Experimental conditions	Conversion	By-products	Remarks	Ref.
Hexane, Octane, Ethanol, 1- Butanol, Toluene, p-Xylene, Methyl ethyl ketone (MEK) and Acetone (0.25, 0.5, and 1 ppm)	TiO ₂ ; Support: FGFs and CCFs	T= 20-25 °C; RH= 15-45%; AFR= 41-255 m³/h; Lamps: UVC and VUV; Intensity= 1.6-4.3 mWcm ⁻²	TiO ₂ /CCFs-VUV and 500 ppb; Hexane: 14%; Octane: 15%; Ethanol: 34%; 1-Butanol: 41%; Toluene: 20%; p-Xylene: 22%; MEK: 29%; and Acetone: 25%.	TiO ₂ /FGFs or CCFs+VUV: • Hexane: formaldehyde, acetaldehyde, 2-butenal, propanal, butanal, tolualdehyde, and hexanal. • Octane: formaldehyde, acetaldehyde, acetone, propanal, butanal and pentanal • Ethanol: formaldehyde, acetaldehyde, 2-butenal, and propanal. • 1-Butanol: formaldehyde, acetaldehyde, 2-butenal, propanal, and butanal. • Toluene: formaldehyde, acetaldehyde, acetone, 2-butenal, butanal, and benzaldehyde. • MEK: formaldehyde, acetaldehyde, acetone, and hexanal. • Acetone: formaldehyde and acetaldehyde.	 Single-pass removal efficiency was varied as: TiO₂/CCFs+VUV>TiO₂/FGFs+VUV>TiO₂/FG Fs+UVC. The lower initial concentrations led to the higher conversion efficiencies. 	[7]
Ethylene, Propylene, and Toluene (200 ppm)	TiO ₂ (TNPs)	T=25 °C; RH=60%; RT=220 min; Lamps= Combination of UVA (13.6 W) and UVB (3 W)	Ethylene: about 73%; Propylene: about 37%; and Toluene: about 61%	For Propylene: • polyethylene and polypropylene	 Calcination at 400 °C for 3 h were the optimum calcination temperature and time for TNP catalyst The surface area of the optimized photocatalyst was 3 times higher than Degussa P25 catalyst (53 m²/g). An increase in calcination time from 1 to 3 h enhanced the surface area as a result of the absence of impurities and water molecules. The required surface area for the total adsorbed of toluene is 19.425 m²/g which is higher than the available 15 m²/g. This means multilayer adsorption happened during the tested conditions. 	[41]

Ethylene, Propylene, and Toluene (200 ppm)	TiO ₂ (TNPs)	T=20 to 80 °C; AFR=100 mL/min; Lamps: Combination of UVA (13.6 W) and UVB (3 W)	Ethylene: 66.3%; Propylene: 80.4%; and Toluene: 46%	For Ethylene: • polyethylene and polypropylene. For propylene: • polyethylene and polypropylene. For Toluene: • NA	 Calcination at 400 °C for 3 h were the optimum calcination temperature and time for TNP catalyst The commercial Degussa P25 showed the lower conversion efficiencies than TNPs. The highest conversion was achieved at 20 °C. Moreover, by increasing the temperature from 20 to 80 °C, the conversion was decreased. In the absence of water vapor, the photocatalytic degradation of toluene did not occur. The toluene illustrated the maximum adsorption capacity on TNP. 	[83]
Ethanol (0.25, 0.5, and 1 ppm)	TiO ₂ ; Support: FGFs and CCFs	T=23 °C; RH=15-30%; Lamp: UVC (18.4 W); Intensity= 29-39 mWcm ⁻²	17-38% for FGFs+UVC; 19-43% for FGFs+VUV; and 31-53% for CCFs+VUV	NA	 TiO₂/FGFs indicated that for the polar VOCs, the adsorption coefficients were roughly one order of magnitude higher than nonpolar VOCs for a specific RH level. TiO₂/FGFs filter showed higher adsorption capacity for polar VOCs. TiO₂/CCFs was better than TiO₂/FGFs in the case of BET surface area, adsorption capacity, resistance against water vapor and photocatalytic activity. 	[61]
Toluene (400 ppmv)	Evonik P25, Cristal PC105, Kronos 1077, and Cristal AT-1; Support: Flat glass disk	T=250 °C; RH=40%; Lamp: Iron halogenide lamp (500 W); Intensity = 3 mWcm ⁻²	52% (after 6 hr) for P25; 50% (after 6 hr) for PC105; 46% (after 6 hr) for Kronos 1077; and 46% (after 6 hr) for Cristal AT-	•P-cresol, hydroquinone, benzoic acid, benzaldehyde, benzyl Alcohol, and carbon dioxide.	Micro-sized catalyst shown a higher hydrophilicity/ hydrophobicity ratio	[49]
Toluene (NA)	Nano TiO ₂ (Degussa P25); Support: ACFs	T= 25±0.5 °C; RH=15, 30, 45, and 60%; Lamp: UV (15 W 254 nm)	14.2% (RH=60%)	Detected intermediate: • Benzaldehyde (about 74-89%), benzyl alcohol (about 9-19%), benzoic acid (about 1%), 2-methyl, p-benzoquinone (about 1-3%) and cresol (about 1-2%)	 The BET surface area of the original ACFs and TiO₂/ACF were 1215.4 and 999.6 m²/g, respectively. Increasing the RH from 15 to 60 % caused increasing in oxidation conversion from 9.5 to 14.2 %. Accumulation rate of benzyl alcohol increased with the increase of RH, however, accumulation rate of benzaldehyde decreased. 	[84]

					•The conversion rate of toluene followed the first-order reaction mechanism.	
Benzene (3.3±0.2 ppmv), Toluene (9.5±0.5 ppmv), Ethylbenzene (9.5±0.5 ppmv), and m-Xylene (15±1 ppmv)	TiO ₂ ; Support: Glass; Coating method: Dip coating	T=NA; RH= 0-35230 ppmv; RT=18-145 s; Lamp: UVC (18 W)	At different RH Benzene: 56-100% (RT~116 s), Toluene: 64-98% (RT~74 s), Ethylbenzene: 65-98% (RT~42 s) and m-Xylene: 68-95% (RT~62 s)	•No by-products were detected above the detection limit (150 ppbv) and CO ₂ was the only detected product.	 No conversion of BTEX neither in the case of photolysis (UV irradiation without catalyst) nor catalysis (catalyst without UV irradiation) was observed. The PCO of toluene and benzene were improved by increasing the water vapor concentration up to the optimum amount, however, the conversions were slightly decreased at more than that concentration. The increase in residence time led to increase in the conversion. 	[56]
Benzene (21.2 ppmv), Toluene (8.57 ppmv), Ethylbenzene (19 ppmv), m-Xylene (10.20 ppmv), o-Xylene (7.53 ppmv), and p-Xylene (23.58 ppmv)	${ m TiO_2}$	T= 50 ± 2 °C; RH=No humidity; Lamp: Low- pressure Hg (12 W)	Benzene: 82%; Toluene: 97%; Ethylbenzene: 83%; m-Xylene: 92%; o-Xylene: 85% and p-Xylene: 98%	 No gas-phase intermediates were detected in the case of benzene, toluene, and ethylbenzene. Ethylbenzene was produced as byproduct in high concentration for m-Xylene while in the case of o,p-Xylene was produced in low concentration. 	• Adsorption on the catalyst surface for aromatic compounds was in the following order: Toluene>m-Xylene>p-Xylene>o- Xylene>Benzene> Ethylbenzene	[85]
Benzene (~47 ppm), Toluene (~42 ppm), Ethylbenzene (~39 ppm), p,m-Xylene (~41ppm), and o-Xylene (~27 ppm)	TiO ₂ ; Support: glass	T= 30.1 ± 0.7 °C; RH=42.2 ± 3.2%; RT= 7.0 ± 0.25 min; Lamps: UVC (8,11,16 W, 254 nm)	Benzene: 82.8%; Toluene: 35.8%; Ethylbenzene: 97.9%; p,m-Xylene: 58.9% and o-Xylene: 67%	•High formation of intermediates for toluene degradation such as benzaldehyde and benzoic acid led to decrease degradation rate constant much faster than the other considered VOCs.	 The particle size of TiO₂ was 10.2±1.43 nm and uniformly dispersed on the surface of the catalyst. By decreasing the particle size specifically under the 30 nm, the photocatalytic activity of TiO₂ and the number of active surface sites increased. The BET surface area of the catalyst powder was 9 m²/g. 	[28]
Benzene (25ppmv)	TiO ₂ ; Support: Zeolite (ZSM-5 with Si/Al of 200 and 470, MCM-41);	T= NA; RH=50%; AFR= 1 L/min; Lamps: Two VUV (4 W), Two UVC	100% for TiO ₂ /ZSM-5 (250 min) and 95% (360 min) 100% for TiO ₂ /MCM-41 (180 min) and 87% (360 min)	NA	• No considerable rutile phase was detected in the XRD pattern of TiO ₂ /MCM-41. However, the small diffraction peak of rutile phase was observed on TiO ₂ /ZSM-5 (200) and TiO ₂ /ZSM-5 (470).	[77]

		(4 W, 254 nm)			 The stable benzene degradation efficiency for the various catalysts followed the order: TiO₂/ZSM-5(200)>TiO₂/ZSM-5(470)>TiO₂/MCM-41>P25. TiO₂/ZSM-5(200) displayed the highest CO₂ concentration which is represented that it was a reliable catalyst for ozone degradation. All the supported TiO₂ catalysts demonstrated higher photoactivity for benzene removal. 	
Acetone and Acetaldehyde (400 ppmv)	Evonik P25, Cristal PC105, Kronos 1077, and Cristal AT-1; Support: Flat glass disk	T=250 °C; RH=40%; Lamp: Iron halogenide lamp (500 W); Intensity= 3 mWcm ⁻²	Acetone: 100% (after 70 min) for P25, 100% (after 60 min) for PC105, 100% (after 90 min) for Kronos 1077, and 49% (after 120 min) for Cristal AT-1 Acetaldehyde: 100% (after 60min) for P25, 100% (after 50min) for PC105, 100% (after 70min) for Kronos 1077, and 46% (after 120 min) for Cristal AT-1	Acetone: No by-product in the case of Evonik P25, Cristal PC105 and Kronos 1077. By-products for Cristal AT-1 were acetaldehyde, acetic acid, formaldehyde, formic acid, and carbon dioxide. Acetaldehyde: No by-product in the case of Evonik P25, Cristal PC105 and Kronos 1077.	• Evonik P25 and PC105 presented high	[49]

T: Temperature

RH: Relative Humidity AFR: Air Flow Rate RT: Residence Time NA: Not Available

TNP: Titanium nanoparticle FGFs: Fiberglass fibers CCFs: Carbon cloth fibers ACFs: Activated carbon fibers

*Table 3. Common preparation method of modified TiO*₂ *photocatalyst for PCO.*

	Synthesize Method	Photocatalyst	Precursor	Dopant/source	Ratio (dopant/TiO ₂)	Treatment Temp./Duration	Cal. Temp./ Duration	Ref.
Metal	Sol-gel	Mn-doped,	Tetra butyl titanate	Mn, Cu, Co, Ni, and Fe	1%	120 °C /6 h	550 °C /4 h	[79]

doping		Cu-doped, Co-doped, Fe-doped, Ni-doped		/Acetate of transition metals				
		Fe-doped	Tetra butyl titanium	Fe/ Iron (III) nitrate	Fe/Ti atomic ratio: 0.1% to 1.5%	80 °C /48 h	480 °C /3 h	[73]
		Fe-doped	Tetra-n-butyl titanium	Fe/ Iron (III) nitrate	Fe/Ti ratios: 0.005, 0.01, 0.05, 0.1, and 0.5	Room Temp. /3 h	400 °C /1 h	[86]
		Fe-doped	Titanium tetraiso- propoxide	Fe/ Iron (III) acetylacetonate	Fe/Ti molar ratios: 0.001 to 0.1%	350–450 °C /4 h	NA	[90]
		Pt-doped	Titanium tetraisopropoxide	Pt/ Hexachloroplatinic acid	0.5 and 1 wt.%	110 °C / overnight	650 °C /2 h	[91]
		Ni-doped	Titanium butoxide	Ni/ Nickel (II) chloride	0.1, 0.3, and 0.5 wt.%	110 °C /12 h	500 °C /5 h	[21]
	Hydrothermal	Ce-doped, Fe-doped, Ni-doped	Tetra butyl titanate	Ce/ Cerium (III) nitrate Fe/ Iron (III) nitrate Ni/ Ni (II) nitrate	0.5 and 5 wt.%	80 °C /24 h	500 °C /2 h	[92]
	Impregnation	Mn-doped	Titanium (IV) butoxide	Mn/ Manganese nitrate	0.5 to 10 wt.%	80 °C / overnight	500 °C /3 h	[62]
		Pt-doped	P25 Degussa	Pt/ Chloroplatinic acid	0.1 and 1 wt.%	120 °C /4 h	400 °C /4 h	[93]
	Hydrolysis	Pt-doped	Titanium tetraiso- propoxide	Pt/ Chloroplatinic acid	0.5, 1 and 2 wt.%	NA	650 °C /2 h	[94]
	Trydrorysis	Ru-doped	Titanium oxosulphate	Ru/ Ruthenium chloride	0.005, 0.01, 0.05, 0.1, 0.25, 0.5, and 0.75 g	100 °C /8 h	600 °C /2 h	[95]
	Thermal assisted microwave	Ir-doped	Titanium (IV) ethoxide	Ir/ Iridium (III) chloride	0.25, 0.5, 0.75, and 1.5 wt.%	60 °C /24 h	200 °C /7 min. 500 W	[96]
		B-doped	Titanium butoxide	B/ Boric acid	1, 3, and 5 wt.%	110 °C /12 h	500 °C /5h	[23]
	Sol-gel	C-doped	Tetra-titanium- isopropoxide	C/ Carbon particle	NA	Room Temp. /12 h	400 °C /1 h	[97]
Non- metal doping	TT 1	C-doped	Titanium (IV) Sulfate	C/ Glaucous	C/Ti molar ratio: 3:50, 1:50, and 1:150	160 °C /12 h	NA	[38]
	Hydrothermal	C-doped	Titanium (IV) Sulfate	C/ Sucrose	1:25	180 °C /12 h	NA	[98]
		C-doped	Ammonium hexafluorotitanate	C/ Glaucous	NA	180 °C /24 h	500 °C	[99]
		N-doped	Tetra butyl titanate	N/ ammonia	NA	NA	600 °C /2 h	[19]

	Thermal treatment	N-doped	P25 powder	N/ Ammonium hydroxide	NA	NA	400 °C /3 h	[100]
	Microemulsion	S-doped	Titanium (IV) isopropoxide	S/ Ammonium sulfate	S/Ti molar ratio: 1, 5, and 20%	110 °C /24 h	600 °C /2 h	[87]
	Hydrolysis	S-doped	Degussa P25	S/ Thiourea	Si/Ti ratio: 0.2, 0.4, 0.8, and 1.6	85 °C /20 h	450 °C /4 h	[101]
	Impregnation	F-doped	P25 powder	F/ ammonium fluoride	mass ratios of 0.3 NH ₄ F/ TiO ₂	85 °C /8 h	500 °C /3 h	[100]
	Solvothermal	S-doped	Titanium isopropoxide	S/Thiourea	molar ratio Ti/S of 1:4	Room Temp. /2 days	350 °C	[102]
	Sol-gel	B, Fe co-doped	Titanium butoxide	B/ Boric acid Fe/ Iron (III) acetylacetonate	B: 3 wt.% Fe:0.5, 1, 3, and 5 wt.%	110 °C /12 h	500 °C /5 h	[23]
	Soi-gei	F,N co-doped	Titanium tetraisopropoxide	F/ ammonium fluoride N/ ammonium hydroxide	NA	85 °C /8 h	500 °C /3 h	[100]
Co- doping	Impregnation	Na-doped Pd/TiO ₂	Degussa P25	Pd/ Palladium (II) nitrate Na/ Sodium nitrate	Pd:1 wt.% Na: 1, 2, 4, and 8 wt.%	110 °C / overnight	400 °C /2 h	[103]
	Microemulsion	Fe, S co-doped	Titanium (IV) isopropoxide	S/ Ammonium sulfate Fe/ Iron(III) nitrate	S/Ti molar ratio:1, 5, and 20% Fe/Ti molar ratio:1.5%	110 °C /24 h	600 °C /2 h	[87]
	Hydrolysis	C, N, and S tri- doped	Titanium isopropoxide	Thiourea	molar ratio of thiourea to TiO_2 is 0, 1, 2, 3 and 6	80 °C /10 h	500 °C /3 h	[104]
		N doped TiO ₂ - Nb ₂ O ₅ composit	Titanium isopropoxide	N/ Ammonium hydroxide Nb ₂ O ₅ / potassium niobate	N: 0.49 at.% TiO ₂ : 50 wt.% Nb ₂ O ₅ : 50 wt.%	80 °C	380 and 500 °C /5 h	[25]
Commo	Sol-gel	SiO ₂ -TiO ₂ Composite	Titanium n- butyloxide	Si/ Tetraethoxysilane (TEOS)	Si/Ti molar ratio:1:100, 5:100, 10:100, 15:100, and 20:100	130 °C /8 h	500 °C	[37]
Compo site		ZnO-TiO ₂ Composite	Titanium isopropoxide	ZnO/ Zinc acetate	50 wt % of ZnO and 50 wt.% of TiO ₂	80 °C /48 h	380 and 500 °C /5 h	[105]
	Impregnation	Ni–TiO ₂ –SnO ₂	Titanium butoxide	Ni/ Nickel (II) chloride SnO ₂ / Triphenyltin hydroxide	Ni: 0.1, 0.3, and 0.5 wt.%	Room Temp/24 h	500 °C /5 h	[21]
	mpregnation	N-TiO ₂ /WO ₃	Titanium tetraisopropoxide	N/Urea WO ₃ / Ammonium paratungstate	N/ WO ₃ / TiO ₂ compositions: 2.5/ 0.5/ 100, 2.5/1.0/	110 °C /2 h	350 °C /2 h	[106]

		pentahydrate	100, 2.5/2.0/ 100,		
			2.5/4.0/ 100, and		
			2.5/6.0/ 100		

Cal.: Calcination; Temp.: Temperature;

Available. Not Table 4. Summary of the common noble and transition metal dopants in the TiO₂ photocatalyst with their operated photocatalytic

conditions and maximum efficiencies.

Contaminant (Concentrati on)	Photocatalyst	Dopant	Experimenta 1 condition	Efficiency before doping	Efficiency after doping	Band gap	Remarks	Ref.
BTEX (0-22 ppmv)	Pt- doped P25, Fe-doped P25, Ce-doped P25; Support: glass; Method: Wet impregnation	Pt, Fe, and Ce: 0.25 % (w/w)	RT=11.5±0. 5 s; Lamp: Low- pressure Hg, UVC, 11 W (253.7 nm)	Benzene: 10-28%; Toluene: 2-12%; Ethylbenzen e: 4-59%; and m-Xylene: 6-87%	Benzene: 4.6-6.5% for Pt/P25, 4.6-12% for Fe/P25, and 17-79% for Ce/P25; Toluene: 3-9% for Pt/P25, 2.7-6% for Fe/P25, and 3-9% for Ce/P25; Ethylbenzene: 15-89% for Pt/P25, 5-80% for Fe/P25, and 10-92% for Ce/P25; m-Xylene: 9-83% for Pt/P25, 6-81% for Fe/P25, and 9-86% for Ce/P25	NA	 In the case of PCO of benzene, only Ce/P25 catalyst was efficient. PCO rates over P25/Ce were higher by a factor of 1.5-10 than those over P25. For PCO of toluene, P25 catalyst proved to be the most active. For PCO of ethylbenzene, all doped catalysts followed the same trend. For all ethylbenzene concentration range, the P25/Ce catalyst was superior to others. In the case of m-xylene, there was no catalyst superior to the others at all concentration range. The PCO conversions of ethylbenzene and m-xylene indicated a great dependence on the initial concentration of the target compound. 	[113]
BTEX (0.1 ppm)	Fe-doped TiO ₂ ; Support: Glass Fiber; Synthesize method: Sol- gel;	Fe-to Ti ratio: 0.005, 0.01, 0.05, 0.1, and 0.5	T=NA; RH=45%; AFR=1 L/min; Lamp: UVA-visible	Under visible light: 0% for Benzene; 5% for	Visible light and 0.01-GF-Fe-TiO ₂ Benzene: 4%; Toluene: 33%; Ethylbenzene: 51% and	NA	 The calculated diameter of TiO₂ crystal was 21.5 nm which was similar to the calculated diameters of the Fe-TiO₂ composites (20.9–21.7 nm). The GF-Fe-TiO₂ composites showed higher photocatalytic activity for BTEX rather than the GF-TiO₂ (reference) photocatalyst. 	[86]

	Coating method:		(daylight and black light	Toluene; 16% for	o-Xylene: 74%		•The photocatalytic activities of BTEX for GF-Fe- TiO ₂ under visible light reduced gradually by	
	Dip-coating		lamp); Visible light intensity= 2.7 mWcm ⁻² and UV light intensity= 0.4 mWcm ⁻²	Ethylbenzene and 29% for o-Xylene Under UV light: 68% for Benzene, 90% for Toluene, 96% for Ethylbenzen e, and 97% for o-Xylene	UV light and 0.5-GF-Fe-TiO ₂ Benzene: 15%; Toluene: 63%; Ethylbenzene:79% and o-Xylene: 86%		 increasing the Fe-to-Ti ratio from 0.01 to 0.50. The BTEX photocatalytic efficiencies of GF-Fe-TiO₂ under UV were lower than those obtained for GF-TiO₂. The GF-TiO₂ photocatalyst demonstrates an absorption edge at λ≈ 410 nm. 	
Benzene (50 ppmv)	R-doped TiO ₂ (R: Mn, Co, Cu, Fe, and Ni), undoped TiO ₂ and P25; Synthesize method: Sol-gel	Mn, Co, Cu, Fe, and Ni 1wt.%	T=NA; RH=50%; AFR=1 L/min; RT=30 s; Lamps: UVC and VUV, 4 W	45.7% for undoped TiO ₂ and 50.2% for P25	58% for Mn/TiO ₂ ; 51.5% for Co/TiO ₂ ; 50.9% for Cu/TiO ₂ ; 45.3% for Fe/TiO ₂ and 50.3% for Ni/TiO ₂ (Under VUV)	NA	 BET surface area of pure TiO₂ was 96.6 m²/g while it declined to less than 50 m²/g for all doped catalysts. It is possible that the presence of transition metals blocked the pores of TiO₂ and increase the growing of particles which lead to the reduction of specific surface area. 	[79]
Benzene (100 ppm)	Cu-doped TiO ₂ ; Support: Porous polyurethane (PU)	Cu: 2, 4, 6, 8, and 10 wt.%	T= NA; RH=40, 60, 80%; AFR= 100 mL/min; Lamps= two 20 W light bulbs (400- 700 nm); Intensity= 50 mWcm ⁻²	NA	86.3% for 6 wt.%- Cu-TiO ₂ /PU	3.20 eV for TiO ₂ /PU; 3.05 eV for 2 wt.% -Cu- TiO ₂ /PU; 2.92 eV for 4 wt.% -Cu- TiO ₂ /PU; 2.85 eV for 6 wt.% -Cu- TiO ₂ /PU; 2.79 eV for 8 wt.% -Cu- TiO ₂ /PU and 2.75 eV for 10 wt.%-Cu- TiO ₂ /PU	 Optimal humidity conditions for the photocatalytic oxidation of benzene in Cu—TiO₂/PU photocatalyst, was 60% RH. Water molecules played an important role in the photocatalytic oxidation process of benzene. The Cu⁺/Cu²⁺ ratios, the benzene removal efficiency and the degree of mineralization enhanced with the increase in the Cu/TiO₂ ratios reached up to 6 wt%, and then gradually decreased with the further increase in ratios (8 and 10 wt%). 	[63]
Toluene (370 ppmv)	Fe-doped TiO ₂ ; Support: glass fiber;	Fe (Fe/Ti atom	T=Room temperature; RH=60%;	Efficiency NA	Efficiency NA 40-96.5%	3.18 eV for 0.1%-Fe- TiO ₂ ;	 No degradation of toluene was observed on the surface of the catalyst in absence of irradiation. Only about 5.8% of the toluene was converted 	[73]

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	Synthesize method: Sol gel	ratio: 0.1- 1.5%)	Lamp: UV- visible, 300 W (UV cut- off filter)	14.9% mineralizatio n for TiO ₂	mineralization (Maximum for Fe/Ti ratio of 0.7%)	3.12 eV for 0.3% Fe- TiO ₂ ; 3.08 eV for 0.5% Fe- TiO ₂ ; 3.02 eV for 0.7% Fe- TiO ₂ ; 3.00 eV for 1.0% Fe- TiO ₂ and 2.96 eV for 1.5% Fe- TiO ₂	after 2h irradiation without the photocatalyst. •By increasing Fe concentration to reach Fe/Ti ratio of 0.7%, the degradation rate of toluene increased significantly and reached the maximum level. Further Fe concentration led to a gradual decrease of the degradation rate. •The Fe ion could replace with Ti ⁴⁺ ions in TiO ₂ lattice which leads to extend the photoactivity of TiO ₂ to visible range and increase the electron and hole pairs' formation.	
Toluene (120 ppm)	Ag-doped TiO ₂ nanotube	Ag 1 wt.%	T=NA; RH=NA; Lamps: UV (8W) and visible light (500W Xe); UV light intensity= 0.87 mWcm ⁻² Visible light intensity= 0.87 mWcm ⁻²	79% for P25 (after 6 h) 90% for pure TiO ₂ under UV	98% (after 4 h) under UV	NA	 By doping Ag into the TiO₂ nanotubes, the density of surface defects was increased and acts as rutile nucleation sites and result in promoting of phase transfer. The concentration of surface oxygen vacancy of anatase particles was improved in the Ag-doping TiO₂ nanotubes, which caused Ag dopant could promote the transformation of anatase to rutile phase at lower calcination temperature. The surface area of 1% Ag-doping TiO₂ nanotubes was 78.5 m² g⁻¹ and it was higher than P25 (52 m² g⁻¹). The pore size distribution of the catalyst was in the range of 1.0 and 30 nm, however, the main pore volume was found in the mesopores with diameters varying from 1.4 to 20 nm. 	[114]
Toluene (200 ppm)	V-doped TiO ₂	V: 0, 2, 4, 6, 8 and 10 wt.%	T=NA; RH=50%; AFR= 200 mL/min; Lamp: visible light, 20 W (400- 700 nm); Intensity= 50 mWcm ⁻²	About 2% for undoped TiO ₂	80% for 6 wt.% V-TiO ₂	3.20 eV for TiO ₂ /PU; 3.09 eV for 2 wt.% -V- TiO ₂ /PU; 2.93 eV for 4 wt.% -V- TiO ₂ /PU; 2.83 eV for 6 wt.% -V- TiO ₂ /PU; 2.76 eV for 8 wt.% -V- TiO ₂ /PU and 2.70 eV for 10 wt.%-V- TiO ₂ /PU	 The optimum value for V content in V/TiO₂ catalyst which showed the highest photocatalytic activity and high mineralization was 6 wt%. The band gap energy of V-TiO₂/PU was gradually decreased with increase in the V/TiO₂ ratios. 	[109]
p-Xylene (80 ppm),	Ni doped TiO ₂ and Degussa	Ni: 0, 1.32,	T=NA; RH=30-	p-Xylene: 33%,	p-Xylene: about 20-60%;	NA	• The concentration of xylene did not change with photolysis (under UV irradiation without	[64]

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m-Xylene	P25;	3.64,	90%;	m-Xylene:	m-Xylene:		photocatalyst) or catalysis (under dark conditions	
(70 ppm),	Support:	4.23,	RT= 54 s;	35%, and	about 18-58%;		in the presence of the photocatalyst).	
and o-	Granular silicate	4.87,	Lamp: UVC,	o-Xylene:	and		• The photocatalytic degradation of xylene over	
Xylene	glass;	4.46,	10 W (254	32%	o-Xylene:		TiO ₂ can be improved by doping Ni at a	
(60 ppm)	Coating	4.57,	nm)		about 18-54%		concentration lower than 4.2 wt.%, otherwise, the	
	method:	and 4.74					photocatalytic activity of modified TiO ₂ would be	
	impregnation	wt.%					worse than that of pure TiO_2 .	
m-Xylene (207 ppb)	Mn-doped TiO ₂	Mn: 0.5, 1, 2, 5, and 10 wt%	T=100 °C; RH=23-25% and 75-80%; RT=0.55- 3.11s; AFR=12 L/min;	33.5±4.8%	2wt.%-Mn-TiO ₂ 50±3.3 %	3.2 eV for TiO ₂ ; 3.1 eV for 0.5 wt.%-Mn-TiO ₂ ; 3.06 eV for 1 wt.%-Mn-TiO ₂ ; 3.01 eV for 2 wt.%-Mn-TiO ₂ ; 2.8 eV for 5 wt.%-Mn-	 The loaded Mn provided more active sites on the photocatalyst surface, thus increased the photocatalytic activity. Mn-doped TiO₂ results in narrowing the band gap. Mn dopants ions caused to reduce the charge carrier recombination rate and improved the separation rate of electron-hole pairs. 	[62]
			Lamp: UV lamp			TiO ₂ and 1.2 eV for 10 wt.%-Mn- TiO ₂	• Removal efficiency increased by loading Mn up to 2 wt%. However, an excess amount of Mn on the surface of catalyst led to decreeing in removal efficiency.	
Isopropanol (1100 ppm)	Rare earth-doped TiO ₂ (RE= cerium, lanthanum, europium and yttrium), Undoped P25; Method: Thermal treatment (Cal. Temp: 900 and 1000 °C	Ce, La, Eu, and Y: 0.01 and 0.025 mol	T=23±2 °C; RH=25- 30%; Lamp: Xenon lamp, 300 W (400- 800 nm); Intensity= 16 mWcm ⁻²	Efficiency NA Acetone formation (ppm h ⁻¹) at Cal. Temp:900 For P25= 15	Efficiency NA Acetone formation (ppm h ⁻¹) at Cal. Temp:900 for La (0.01 mol)=26; La (0.025 mol)=32; Eu (0.01 mol)=27; Eu (0.025 mol)=19; Y (0.01 mol)=18; Y (0.025 mol)=21; Ce (0.01 mol)<0.5 and Ce (0.01 mol)<0.5	3.01 eV for P25; 3.04 eV for La (0.01 mol); 3.04 eV for La (0.025 mol); 3.03 eV for Eu (0.01 mol); 3.04 eV for Eu (0.025 mol); 3.03 eV for Y (0.01 mol); 3.03 eV for Y (0.025 mol); 2.91 eV for Ce (0.01 mol) and 2.90 eV for Ce (0.025 mol)	 The thermal treatment of the undoped P25 powder, caused to a decrease of its photocatalytic activity due to the concomitant size growth of the titania particles, and the decrease of its specific surface area. The three catalysts La (0.025 mol)-900, Eu (0.01 mol)-900 and La (0.01 mol)-900 all had high photocatalytic activity under visible-light irradiation. The rare earth doped TiO₂ modifications led to attach more hydroxyl groups to the surface of the photocatalyst and generate chemical oxidative species, such as hydroxyl radicals which enhanced the photocatalytic activity. 	[124]
Ethanol (0.2 vol. %) (Concentrati on of 0.2 vol.%, in	Pt-doped TiO ₂	Pt: 0.5 and 1 wt.%	T=60,80, and 100 °C; RH=NA; AFR=20-70 L/h; Lamps= two	For undoped TiO ₂ : 69%, 82% and 80% (at 60, 80, and 100 °C,	98% for 0.5wt.%- Pt/TiO ₂ (at 80 °C) and 80% for 1 wt.% Pt/TiO ₂ (at 80 °C)	3.2 eV for TiO ₂ ; 2.8 eV for 0.5 wt.%-Pt- TiO ₂ and 3.0 eV for 1 wt.%-Pt- TiO ₂	 No ethanol conversion was observed in dark adsorption for all tested temperature (60, 80 or 100 °C). The conversion of ethanol enhanced when the reaction temperature increased to 80 °C and then decreased for the reaction temperature of 100 °C. 	[91]

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helium flow with an oxygen/etha nol ratio of 2)			UVA-LEDs modules (Each module consisted of 40 UV- LEDs, 365nm); Intensity= 90 mWcm ⁻²	respectively)			 Ethanol conversion achieved with the catalysts with 1 wt.% Pt was slightly lower than those obtained with 0.5 wt.% Pt. Acetaldehyde was the main reaction product observed in all tests. 	
Acetonitrile (500 ppm)	M-doped TiO ₂ M: V, Cr, Fe, Co, Mn, Mo, Ni, Cu, Y, Ce, and Zr	Ti/M=20 atomic ratio (M: V, Cr, Fe, Co, Mn, Mo, Ni, Cu, Y, Ce, and Zr)	T=30-33 °C; RH=NA; AFR=100, 200, 300, and 500 mL/min; Lamps: Six fluorescent white light lamps, 8W	4.5% for P25	70% for Cr-TiO ₂ ; 21% for Fe-TiO ₂ ; 3.3% for Co-TiO ₂ ; 3.5% for Ce-TiO ₂ ; 5% for Cu-TiO ₂ ; 16.5% for V-TiO ₂ ; 2% for Ni-TiO ₂ ; 3.5% for Mn-TiO ₂ ; 4.5% for Zr-TiO ₂ and 3% for Y-TiO ₂	3.11 eV for P25; 2.82 eV for Cr-TiO ₂ ; 2.69 eV for Fe-TiO ₂ ; 2.54 eV for Co-TiO ₂ ; 3.13 eV for Ce-TiO ₂ ; 2.86 eV for Cu-TiO ₂ ; 2.63 eV for V-TiO ₂ ; 2.86 eV for Mn-TiO ₂ ; 3.19 eV for Mo-TiO ₂ ; 3.21 eV for Zr-TiO ₂ and 3.20 eV for Y-TiO ₂	 The Cr, Fe and V-doped TiO₂ nanoparticles showed better photocatalytic efficiency under the visible light. The incorporation of the other transition metals including Mn, Mo, Ce, Co, Cu, Ni, Y, and Zr demonstrated an inhibition effect on the photocatalytic activity. Cr-doped TiO₂ exhibited a superior photocatalytic activity and the rate constant for this modified catalyst was about 8–19 times higher than the other tested metal doped catalysts. 	[125]
Acetaldehyde (500 ppm) and Toluene (100 ppm)	Ir-doped TiO ₂ ; Support: glass dish; Coating method: Spray Coating	Ir: 0.25, 0.5, 0.75, and 1.5 wt.%	T= Room temperature; Lamp: LED (455 ± 0.1 nm); Intensity= 1 mWcm ⁻²	Efficiency NA About 270 CO ₂ ppm for undoped TiO ₂	Efficiency NA $\begin{bmatrix} 500 \text{ ppm CO}_2 \text{ for} \\ 0.5 \text{ wt% Ir-doped} \\ \text{TiO}_2 \end{bmatrix}$	3.14 eV for bare TiO ₂ (brookite) and 1.88 eV for 0.5wt.%-Ir-TiO ₂	 The specific surface areas for 0.25, 0.5, 0.75 and 1.5 wt% Ir-doped TiO₂ brookite corresponded to 172.13, 208.41, 219.67, and 231.15 m²g⁻¹ respectively, and for the undoped brookite was 128.18 m²g⁻¹. 0.5 wt% Ir-doped TiO₂ had the highest photocatalytic activity for decomposition of both acetaldehyde and toluene. The band gaps of the undoped and 0.5wt% Ir-doped TiO₂ catalyst were 3.14 eV and 1.88 eV respectively. 	[96]
Acetone (NA)	Ru-doped TiO ₂	Ru: 0.75, 0.5, 0.25, 0.1, 0.05, 0.01 and 0.005 g	Lamps: UV (365 nm and 400 nm)	Rate constant for P25 Degussa was 0.00041	Efficiency NA Rate constant for 0.05-Ru-TiO ₂ was 0.00059 and for 0.01-Ru-TiO ₂ was 0.00058	3.2 eV for TiO ₂ ; 2.7 eV for 0.75g-Ru- TiO ₂ ; 2.9 eV for 0.5g-Ru-TiO ₂ ; 2.9 eV for 0.25g-Ru- TiO ₂ ; 2.9 eV for 0.1g-Ru-TiO ₂ ;	 The photocatalytic activities of Ru-doped TiO₂ catalysts under visible light were strongly affected by the decreasing band gap energy. The rate constant (k) of the degradation of acetone under visible light was highest for the catalyst 0.05-Ru-TiO₂. The total pore volume and a pore radius of 	[95]

		3.1 eV for 0.05g-Ru-	catalyst increased with increasing content of Ru
		TiO_2 ;	ions in the TiO_2 lattice.
		3.1 eV for 0.01g-Ru-	• Ru-doped TiO ₂ catalyst structure was
		TiO_2 and	microporous with pore sizes in the range from 2
		3.15 eV for 0.005g-Ru-	to 4 nm.
		TiO_2	

T: Temperature

RH: Relative Humidity AFR: Air Flow Rate RT: Residence Time NA: Not Available

Table 5. Summary of the common non-metal dopants in the TiO_2 photocatalyst with their operated photocatalytic conditions and maximum efficiencies.

Contaminant	Photocatalyst	Dopant	Experimental	Efficiency	Efficiency	Band gap	Remarks	D 6
(Concentration)		F	condition	before doping	after doping	8F		Ref.
Toluene (40 ppm)	C-doped TiO ₂ ; Support: dish	С	T=25±1 °C; RH=60%; Lamp: Xe-lamp, 150 W (λ>425 nm)	About 5%	About 58.5%	3.10 eV for TiO ₂ and 2.30 eV for C- TiO ₂ -200 °C	 C-TiO₂ at post-thermal temperature of 200 °C showed the higher activity. The rate constant of the pseudo-first-order reaction of C-TiO₂-200 was 7.69×10⁻³ min⁻¹ while for undoped TiO₂ was 0.49×10⁻³ min⁻¹. The specific surface area of undoped TiO₂ and C-TiO₂-200 were 118.97 and 79.11 m²/g, respectively. Un-doped TiO₂ only absorbed the UV light, however, C-TiO₂ exhibited absorption of visible light. 	[98]
Toluene (40 ppm)	C-doped TiO ₂ ; Support: dish; Method: Hydrothermal	C 0.9, 2.1, and 3.4 wt.%	T=25±1 °C; RH= Controlled by a CaCl ₂ dryer; Lamp: Visible light, 150 W, (λ>425 nm)	5.2% for undoped TiO ₂ and 8.9% for P25 (after 2 h)	2.1 wt.% C- doped TO ₂ 81.8% (after 2 h)	3.10 eV for TiO ₂ ; 2.97 eV for 0.9 wt.%-C-TiO ₂ ; 2.90 eV for 2.1 wt.%-C-TiO ₂ and 2.78 eV for 3.4 wt.%-C-TiO ₂	 C-doped TiO₂ with 2.1 wt.% showed the maximum photocatalytic activity which considered as the optimum value for carbon doping in this study. C-doping did not change the particles size and the crystal structure of catalysts significantly. The TEM images for C-doped TiO₂ (2.1 wt.%) and undoped TiO₂ showed the mesoporous structure which consists of the huge amount of mono-dispersed particles with a size of nearby 10-12 nm. C-doped TiO₂ catalyst was prepared by green synthesis method. 	[38]

Toluene (100 ppm)	C- doped TiO ₂ ; Support: optical fiber; Method: Solgel and dip coating (1 to 5 times coating, the film thickness of TiO ₂ was 1.6, 2.6, 3.2, 4.0, 4.2 µm)	C 0.2 g carbon particle (with average diameter less than 50 nm)	T=25 °C; RH=30%; AFR= 3 L/min; Lamp: Hg lamp, (300 nm, 405 nm, 436 nm)	35-79% (during 0.5 min to 30 min)	62-94% (during 0.5 min to 30 min)	Calculated band gap: 3.07 and 2.85 eV for N - TiO ₂	 Photodegradation efficiency was enhanced as the TiO₂ thickness film was increased from 1.6 μm to 3.2 μm. However, by increasing thickness over 3.2 μm, the efficiency was decreased as the excessive film layer prevented UV light from passing through the TiO₂ film. The specific surface area for TiO₂ film (thickness 3.2 μm) was 55 m²g⁻¹ and for C-TiO₂ was 230 m²g⁻¹ By-products were formic acid and ethyl ester. 	[97]
Toluene (66.3 ppm)	C-doped TiO ₂ and Degussa P25; Support: dish; Method: Hydrothermal	С	T=25±1 °C; RH=NA; Lamp: Xe lamp, 150 W	Efficiency NA Rate constant k= 0.0008 min ⁻¹	Efficiency NA Rate constant k= 0.0037 min ⁻¹	NA eV for P25; 2.80 eV for C- TiO ₂ -500 °C and 2.90 eV for C- TiO ₂ -600 °C	 The PCO of toluene followed the pseudo-first-order reaction. The average diameter of the TiO₂ hollow spheres after calcination at 500°C for 1 hr was 0.5-3.0 μm. Degussa P25 exhibited lower activity rather than C-doped TiO₂. The superior activity of C-doped TiO₂ at 500 °C resulted of the crystal structure and the mesoporous microstructure. The band gap of the C-doped TiO₂ was narrower than that of the pure TiO₂. 	[99]
BTEX (0.03 to 3.1 ppm)	S-doped TiO ₂ and Degussa P25; Support: glass	S Molar ratio Titanium isopropox ide/thiour ea was 1:4	T=19-25 °C; RH=10-20, 50-60, and 80-90%; AFR=0.5-6.0 L/min; Lamp: Fluorescent Visible light, 8 W (400-700 nm); Intensity= 1.9 mWcm ⁻²	About 5% for Benzene; 14% for Toluene; 13% for Ethylbenzene; 11% for m,p- Xylene and 13% for o- Xylene	About 41- 1.3% for Benzene; about 85-1% for Toluene; about 97-13% for Ethylbenzene; about 97-53% for m,p- Xylene and about 95-51% for o-Xylene (in the range of 0.03 ppm to 3.1 ppm)	Calculated band gap: 3.11-1.73 eV for S -TiO ₂	 By increasing the AFR from 0.5 to 6 L/min, the degradation of BTEX decreased, thus, for the lowest AFR, the decomposition efficiency was maximum. The degradation efficiency of toluene and benzene exhibited a great dependency on RH and the decomposition efficiency of it increased by decreasing the relative humidity. The efficiency of Ethylbenzene and m,p-xylene decomposition exhibited a less dependency on RH and almost equal for both 10-20% and 50-60%. Then the efficiency decreased by increasing the RH to 80-90%. The degradation efficiency of BTEX exhibited an increasing trend with decreasing initial concentration because of competitive adsorption between pollutant molecules on the photocatalyst surface. 	[65]

BTEX (0.1, 0.5, 0.7, and 0.9 ppm)	S-doped TiO ₂ ; Support: Aluminum sheet	S/Ti ratio: 0.2, 0.4, 0.8, and 1.6	T=NA; RH= 45%; AFR=1, 2, 3, and 4 L/min; RT= 2, 2.6, 3.9, and 7.8 s; Lamp: Daylight (8 W); Intensity= 2.8 mWcm ⁻²	TiO ₂ -AS About 2% for Benzene; 11% for Toluene; 21% for Ethylbenzene; and 36% for Xylene	For S-TiO ₂ - 0.8-AS about 34% for Benzene; about 78% for Toluene; about 91% for Ethylbenzene; and about 94% for Xylene	3.02 eV for TiO ₂ and 2.75 eV for S-TiO ₂ -0.8-AS	 The photocatalytic degradation efficiencies of BTEX under visible irradiation enhanced with increasing S/Ti ratios from 0.2 to 0.8, then decreased when the ratio increased to 1.6. The photocatalytic degradation efficiencies of all target compounds over S-TiO₂-0.8-AS decreased with increasing air flow rate. The XRD of both S-TiO₂-0.8-AS and TiO₂-AS samples showed an anatase crystal structure with a major band at 2θ= 25.2° and a rutile crystal structure with a major band at 2θ = 27.4°. The photocatalytic degradation efficiencies showed an increasing pattern with decreasing initial contaminant concentration. 	[101]
BTEX (100 ppb)	N-doped TiO ₂ and Degussa P25; Support: Pyrex	N	T=19-25 °C; RH=10-20, 50-60, and 80-90%; AFR=0.5,1.0, and 2.0 L/min; RT= 22, 11, and 5.5 s; Lamp: Fluorescent visible light, 8 W (400-700 nm); Intensity= 1.9 mWcm ⁻²	About 14% for Benzene; 17% for Toluene; 16% for Ethylbenzene; 12% for m,p-Xylene and 17% for o-xylene	About 24% for Benzene; about 64% for Toluene; about 93% for Ethylbenzene; about 96% for m,p-Xylene and about 91% for o-Xylene (AFR=0.5 L/min, RH=50-60%)	Calculated band gap: 3.11-1.73 eV for N -TiO ₂	 For all three examined flow rates, the PCO degradation efficiency for ethyl benzene and o,m,p-xylenes were higher than that of toluene or benzene. The PCO degradation efficiency of all target compounds increased as the flow rate decreased. Benzene exhibited the lowest degradation. Increasing the distance of the catalyst from the light source led to the decrease in the light intensity in the reactor and caused a drop in the degradation efficiency of the target compounds. 	[39]
Toluene (500 ppm)	N-doped TiO ₂ , undoped TiO ₂ , A-287, PC105, DegussP25, and PC103; Support: Pyrex	N N/Ti molar ratio of 6.6, 7.6 and 9.2	T=NA; RH=60 %; Lamp: Solar lamp, 300 W (380-600 nm); Intensity= 1.2×10 ⁻⁵ Einstein dm ⁻³ s ⁻¹	13.7% for T9 (undoped TiO ₂ at pH=9) 55.8% for A- 287, 47.7% for PC105, and 56% for P25	79.6% for TN9 (prepared at pH=9)	for N -TiO ₂	 The reaction kinetics for all synthesized catalyst could be explained as a pseudo-first-order rate equation. The N-doped TiO₂ (prepared at pH=9 has the fastest kinetics constant which value is k = 6.0×10⁻³ min⁻¹ 	[24]
Acetaldehyde (600–3650 ppm)	N-doped TiO ₂ ; Method: Thermal treatment	N	T=25 °C; RH=NA; AFR=1000 mL/min;	88.1% for 600 ppm; 77.3% for 1000 ppm;	100% for 600 ppm; 92.1% for 1000 ppm;	3.16 eV for P25 and 2.91 eV for N- TiO ₂	• The absorption range of N-doped TiO ₂ photocatalyst was expanded to the visible range.	[100]

			Lamp: High pressure Hg lamp, 80 W (185–580 nm)	66.1% for 1700 ppm; 58.8% for 2200 ppm; and 51.2% for 3650 ppm	87.2% for 1700 ppm; 81.2% for 2200 ppm and 77.6% for 3650 ppm			
Toluene (NA)	B-doped TiO ₂ ; Support: Pyrex; Method: Sol- gel	Br: 1, 3, and 5 wt.%	T=70 °C; RH=NA; Lamp: UVC (10 W, 254 nm) and halogen (150 W,400 nm)	92 %	About 100% for B-doped TiO ₂	Calculated band gap: 3.11 eV for B-TiO ₂	 All samples were pure anatase TiO₂ after calcination and no phase change happened during the doping process. In XRD pattern, the crystallite size reduced with doping B into the TiO₂. The most characteristic feature of B-doping was that both surface area and pore volume of TiO₂ were increased. All reactions followed pseudo first- order kinetics. 	[23]
Acetaldehyde (600–3650 ppm)	F-doped TiO ₂ ; Method: Impregnation	F	T=25 °C; RH=NA; AFR=1000 mL/min; Lamp: High pressure Hg lamp, 80 W (185–580 nm)	88.1% for 600 ppm; 77.3% for 1000 ppm; 66.1% for 1700 ppm; 58.8% for 2200 ppm and 51.2% for 3650 ppm	97% for 600 ppm; 81% for 1000 ppm; 76% for 1700 ppm; 68% for 2200 ppm and 62% for 3650 ppm	3.16 eV for P25 and 3.09 eV for F- TiO ₂	 The F-doped TiO₂ catalyst exhibited a stronger light absorption in the UV region, but did not have any significant photoactivity in visible range. The F-doped TiO₂ reduced the recombination process of charge carrier and led to the formation of new active sites. The F-doped TiO₂ led to increase pollutants adsorption capacity and photocatalytic activity. F-doped TiO₂ photocatalyst had less specific surface area compared to P25, which it indicates that crystals' growth was improved. 	[100]

T: Temperature

RH: Relative Humidity AFR: Air Flow Rate RT: Residence Time NA: Not Available

Table 6. Summary of the common co-doped TiO_2 photocatalyst with their operated photocatalytic conditions and maximum efficiencies.

Contaminant Photocatalyst Dopant	Experimental	Efficiency	Efficiency	Band gap	Remarks	

(Concentration)			condition	before	after doping			Ref.
,				doping	1 &			
Toluene (40 and 133 ppm)	N, C, and S tri-doped TiO ₂ ; Support: dish	The molar ratio of Ti:R 1:0, 1:1, 1:2 and 1:3, (TCNS0, TCNS1, TCNS2 and TCNS3)	T=25±1 °C; RH=NA; Lamp: Xe-lamp, 150 W	Efficiency NA Rate constant, k (min ⁻¹) for P25= 6.95×10 ⁻²	Efficiency NA Rate constant, k (min ⁻¹)= 12.8×10 ⁻²	3.18 eV for TCNS0; 3.0 eV for TCNS1; 2.88 eV for TCNS2 and 2.67 eV for TCNS3	 TCNS1 (R-doped TiO₂, R=1, Ti: R=1:1) showed the higher activity under UV-vis light irradiation. The rate constant of the pseudo-first-order reaction of TCNS0, TCNS1 under UV-vis light were 5.54×10⁻² min⁻¹ and 12.7×10⁻² min⁻¹ respectively. The specific surface area of TCNS0 and TCNS1 were 155.1 and 154.3 m²/g respectively. 	[72]
Formaldehyde (200±10 ppm)	N, C, and S tri-doped TiO ₂ and P25	N, C, and S. The molar ratio of thiourea to TiO ₂ was 0, 1, 2, 3 and 6	T= Ambient temperature; RH=NA; Lamp: daylight (420 nm); Intensity= 0.46 ± 0.01 mWcm ⁻²	2% for undoped TiO ₂ and 5% for P25	32.6% for N, C, and S- doped TiO ₂ -3	3.0 eV for P25; 3.2 eV for TiO ₂ ; 2.8 eV for N-C,S- TiO ₂ -1; 2.5 eV for N-C,S- TiO ₂ -2; 2.4 eV for N-C,S- TiO ₂ -3 and 2.2 eV for N-C,S- TiO ₂ -6	 The tri-doped catalysts could be activated by visible light and more photogenerated charge carriers can be generated and participate in the photocatalytic reactions. The C, N, S tri-doped TiO2 catalyst caused band gap narrowing and reduced the recombination of photo-generated charge carriers and thus enhanced photocatalytic activity. 	[104]
Acetaldehyde (600–3650 ppm)	F and N co- doped; Method: Sol- gel	F and N	T=25 °C; RH=NA; AFR=1000 mL/min; Lamp: High- pressure Hg lamp, 80 W (185–580 nm)	88.1% for 600 ppm; 77.3% for 1000 ppm; 66.1% for 1700 ppm; 58.8% for 2200 ppm and 51.2% for 3650 ppm	100% for 600 ppm; 98.2% for 1000 ppm; 94.1% for 1700 ppm; 87.1% for 2200 ppm and 80.7% for 3650 ppm	3.16 eV for P25 and 2.86 eV for F-N- TiO ₂	 The absorption ranges of F and N co-doped TiO₂ photocatalysts was extended to the visible light. The F and N co-doped TiO₂ catalyst had demonstrated the highest photocatalytic activity than other catalysts. Doping both N and F ions caused synergy effects such a narrowing band gap, formation of new active sites and enhancement of photoactivity. By increasing the initial contaminant concentration in the gas phase, the number of the available active sites of the catalyst became less than the amount of the pollutants. The F and N co-doped TiO₂ photocatalyst had less specific surface area compared to P25, which it indicates that crystals' growth was improved. 	[100]

Hexane and Butyl acetate (100 ppm each) and their mixture	Ag and V co- doped TiO ₂ ; Support: polyurethane	Ag and V (0-6 wt.%)	T=NA; RH=50 %; AFR=100 mL/min; RT= 72s; Lamps: Two white light lamps, 20 W (400<λ<700); Intensity= 50 mWcm ⁻²	NA	Individual: 93.7% for Hexane and 95.5% for Butyl acetate In mixture: 75% for Butyl acetate and 48 % for Hexane	3.20 eV for TiO ₂ /PU; 3.02 eV for 6Ag/0V-TiO ₂ /PU; 2.91 eV for 5Ag/1V-TiO ₂ /PU; 2.86 eV for 4Ag/2V-TiO ₂ /PU; 2.79 eV for 3Ag/3V-TiO ₂ /PU; 2.77 eV for 2Ag/4V-TiO ₂ /PU; 2.81 eV for 1Ag/5V-TiO ₂ /PU and 2.83 eV for 0Ag/6V-TiO ₂ /PU	 The optimum catalyst (4Ag/2V-TiO₂/PU) had combination weight ratios of 4wt% Ag/TiO₂ and 2wt% V/TiO₂ led to the highest BET surface area of 230.5 (m²/g). The surface area of the co-doped photocatalyst was also higher than that of the undoped and single-dopant photocatalysts. Due to the high polarity of butyl acetate, higher amount of butyl acetate was adsorbed on the surface of 4Ag/2V-TiO₂/PU than hexane. 	[108]
Acetone (NA)	C and Mo codoped TiO ₂ and undoped TiO ₂ (Degussa P25); Support: dish	Mo: 2 at. % and C: 1 at. %	T=NA; RH=NA; Lamp: UV lamp, 125 W and glass filter to produce visible light	53%	76.5%	Calculated band gap: 2.96 eV for C, Mo- TiO ₂	 Doping a small amount of C had no obvious effect on the crystalline size, while, the crystalline size of the Mo-doped and co-doped samples were smaller than that of the undoped one. C and Mo enhanced both the absorption ability of visible light and promoted the separation of photogenerated charge carriers. The surface area and pore size of C, Mo co-doping TiO₂ significantly increase compared with those for P25. 	[138]
Toluene (39.8 and 132.68 ppm)	Fe and N co- doped TiO ₂	Fe: 0, 0.005, 0.02, 0.05, 0.1, 0.5, 1.0 and 2.0 wt.%	T=25±1 °C; RH= 60%; Lamp: Xe-lamp, 150 W	63%	Around 97%	Calculated band gap: 2.92 eV for Fe, N- TiO ₂	 The optimum concentration of Fe ions was 0.02wt.% for the maximum photodegradation of toluene. The Fe, N co-doped TiO₂ catalyst demonstrated about three times higher photoactivity than that of N-TiO₂ nanocrystals. The reused Fe/N-TiO₂ photocatalyst did not demonstrate an apparent change in photocatalytic activity. 	[120]
Toluene (NA)	B and Fe co- doped TiO ₂ ; Method: Sol- gel	B: 3 wt.% Fe: 0, 0.5, 1, 3, and 5 wt.%	T=70 °C; RH=NA; Lamps: UVC (10 W, 254 nm) and halogen	Under Visible light About 8.8% (after	Under Visible light: About 87.9% (after 180 min)	Calculated band gap: 3.11-2.07 eV for B, Fe-TiO ₂	 All samples were pure anatase TiO₂ after calcination and no phase change happened during the co-doping process. In XRD pattern, the crystallite size reduced with 	[23]

			lamp (150, 400 nm); Visible light emitting flux: 3.9×10^{-6} Einstein/s and UV light emitting flux: 1.47×10^{-6} Einstein/s	180 min)			doping B and Fe ions into the TiO ₂ and the decrease of crystallite size was more result of B-doping than with Fe-doping. • All reactions follow pseudo first- order kinetics rate.	
Toluene (800 ppmv)	S and Fe co- doped TiO ₂ ; Support: Pyrex	Fe: 1.5% S/Ti molar ration: 1% to 20%	T=Room temperature; RH=75%; AFR= 100 mL/min; Lamps: UVA and simulating sunlight, 6W for both	Efficiency NA Reaction rate for P25 (mol/s.m²): 3.87634× 10 ⁻¹⁰	Efficiency NA Reaction rate for S, Fe codoped TiO ₂ (mol/s.m ²): 4.77957× 10 ⁻¹⁰	NA	 The S, Fe co-doped TiO₂ catalyst has improved photoactivity compared to the single-doped Fe-TiO₂ and P25 under both sunlight and UV-light irradiation. The XRD patterns of both S, Fe co-doped, and S-doped TiO₂ were included 100% anatase crystal phase. In the S, Fe co-doped TiO₂ catalysts, adding the S content led to a gradual increase of the BET surface area value and pore size. All S, Fe co-doped TiO₂ catalysts showed great absorption in the visible light compared to the S-doped TiO₂ catalysts. Generated by-product was benzaldehyde. 	

T: Temperature

RH: Relative Humidity AFR: Air Flow Rate RT: Residence Time

NA: Not Available

Table 7. Summary of the common semiconductor coupled TiO₂ photocatalyst with their operated photocatalytic conditions and maximum efficiencies.

Contaminant (Concentration)	Photocatalyst	Dopant	Experimental condition	Efficiency before doping	Efficiency after doping	Band gap	Remarks	Ref.
Propylene (NA)	MnCO ₃ -TiO ₂ and P25; Support: glass plate	MnCO ₃ /TiO ₂ ratios: 1:10, 1.5:10, and 2:10	T=NA; RH=NA; RT=8 min; AFR=200 mL/h;	0% for MnCO ₃ , and 7% for P25	70%	Calculated band gap: 2.96 eV for MnCO ₃ -TiO ₂	 MnCO₃-TiO₂ (2:10) exhibited the highest activity for degradation of propylene. The majority of MnCO₃ was nanoparticle (NP) with an average size of 4–5nm. Comparing with P25 and MnCO₃, MnCO₃-TiO₂ 	[139]

			Lamp: Xenon lamp,500 W (λ≥420 nm); Intensity= 25 mWcm ⁻²			represented high absorption in visible light, especially in 400–500 nm region.	
Hexane and Isopropyl alcohol (0.1 ppm for each)	N doped TiO ₂ - WO ₃	N/WO ₃ /TiO ₂ composition s: 2.5/0.5/100, 2.5/1.0/100, 2.5/2.0/100, 2.5/4.0/100, and 2.5/6.0/100	T=NA; RH=45%; AFR=0.1, 0.5, and 1 L/min; RT= 1.3, 0.26, and 0.13 min; Lamp: 8 W, 400-720 nm; Intensity= 0.48 mWcm ⁻²	4% for Hexane	WO ₃ -N- TiO ₂ -4 43.4(±1.1)% for pure Hexane, and 72.7(±1.6)% for pure IPA In mixture: 34.2(±1.5)% for Hexane and 72.6(±1.7)% for IPA	2.80 eV for WO ₃ -N-TiO ₂ -2; and led to the lower surface area. The photocatalytic degradation efficiency for pure IPA was higher than that for pure hexane. The photocatalytic degradation efficiency for hexane mixed with IPA, was lower than that for pure hexane, while the IPA decomposition efficiency remained relatively constant.	[106]
Toluene (NA)	Pt-SnO ₂ - R/TiO ₂ - A/TiO ₂ and Titania nanotubes (TNT)	Sn/Ti ratio: 2(S1), 6(S2), 10(S3), 15(S4) at. % and P1 was S3 doped with Pt (1 wt.%)	NA	Efficiency NA CO ₂ production rate: 20.43 ppm/h	Efficiency NA CO ₂ production rate: 113.81 ppm/h	•The anatase phase was observed in all photocatalysts. However, the rutile and SnO ₂ appeared just for S3, S4, and P1 catalysts. •The specific surface area was in the order S0 < S3 < P1, demonstrating that the number of mesoporous structures existing in the photocatalysts increased the specific surface area. •S3 and P1 demonstrated larger pore volumes, pore diameters, more active site and better adsorption than TNT. •By increasing the Sn/Ti ratio to 10 at. % the photocatalytic efficiency was enhanced. However, by more increasing from 10 to 15 at.% the efficiency was decreased.	[22]
Toluene (177 ppm)	xNi- TiO ₂ - SnO ₂ ; Method: Sol- gel and wet- impregnation	x=0.1, 0.3, and 0.5 wt% SnO ₂ : 1 wt%	T=70 °C; RH=NA; AFR=320 cm³/min; Lamp: Halogen	5.7%	51%	• All reactions for all different catalysts followed pseudo first-order kinetics as the rate of reaction for certain	[21]

			lamp, 150W; Visible light emitting flux: 3.9×10^{-6} Einstein s ⁻¹				highest value (0.0058 min ⁻¹) among the tested catalysts. • xNi-doped TiO ₂ was active in visible light while a composite of SnO ₂ onto the TiO ₂ surface further increase the efficiency of these catalysts. • The Ni-doped TiO ₂ nanoparticles host SnO ₂ on their surface caused to reduce the recombination rate. • TiO ₂ and TiO ₂ —SnO ₂ catalyst were inactive under visible light irradiations	
Toluene (7 ppm)	SiO ₂ doped TiO ₂ ; Method: Sol- gel	Si/Ti molar ratio of 1:100, 5:100, 10:100, 15:100, and 20:100	T=NA; Humidity=1.6 g/m³; Lamp: UVA mercury lamp, 300 W (365 nm)	69.6% for undoped TiO ₂ and 70.5% for P25	90.6%	NA	 The 10% SiO₂ doped TiO₂ fibrous catalyst calcined at 400 °C had the highest degradation rate compared to the other catalysts. Based on the results 400°C was selected as the optimum temperature to remove organic compounds. The surface area for SiO₂ doped TiO₂ ratio at 10% was 266 m²/g. 	[37]
Benzene (280 ppm)	Ag ₃ VO ₄ -TiO ₂ And pure TiO ₂ (TiO ₂ at cal. Temp:500 °C); Method: Hydrothermal	mass fraction of Ag ₃ VO ₄ in nanocompos ites was 0.1, 0.5, 1, 2, and 5 %,	T=30±1 °C; RH=NA; AFR= 20 cm³/min; Lamps: Xe-arc lamp, 500 W (450<λ<900), and Xe-arc lamp, 500 W (λ<450)	Under visible light: 8.9% Under solar light: 43%	Under visible light: 40% Under solar light: 94.5%	Calculated band gap: 2.76 eV for Ag ₃ VO ₄ -TiO ₂	 By increasing the amount of Ag₃VO₄ in Ag₃VO₄/TiO₂ nanocomposites, the anatase phase was increased. 0.5% Ag₃VO₄ in nanocomposites catalyst presented optimum activity for benzene degradation under visible light irradiation. The absorption edge and band gap of Ag₃VO₄ were estimated to be 560 nm and 2.2 eV. By increasing the amount of Ag₃VO₄ till 2% the BET specific surface area of Ag₃VO₄/TiO₂ increased, and more increasing of that resulted in decreasing of the surface area. The specific surface area for 0.5% Ag₃VO₄/TiO₂ nanocomposites and TiO₂ were determined to be 48.98 and 28.24 m²g⁻¹, respectively. 	[36]
Benzene (NA)	CdS-TiO ₂ ; Support: fiber glass; Method: Sol- gel	Cd/Ti: 5.65, 7.93, 9.20, and 22.5 %	T=NA; RH=NA; Lamp: High- pressure mercury, 160 W	55.4% for TiO ₂ /FGC	92.8% for 5.65%-CdS / TiO ₂ -FGC under UV- vis and and 32.7% for 5.65%- CdS / TiO ₂ - FGC under	Calculated band gap: 3.11 eV for CdS-TiO ₂	 By increasing the CdS amount, the photocatalytic activity of catalysts initially increased and then showed a downward trend. All the catalysts with different amount of CdS under visible light irradiation showed greater photocatalytic activities than TiO₂-FGC. The recombination rate of charge carriers in CdS-TiO₂ composite was effectively decreased. 	[140]

					visible			
Benzene (31.3-62.6 ppb) and Toluene (26.54-53.07 ppb)	TiO ₂ composite ZnO	50% TiO ₂ and 50% ZnO	T=NA; RH=NA; AFR=0.42 - 2.54 L/min; RT=5-30 min; Lamp: UVA (365-400 nm); Intensity= 0.4, 0.6, and 1 mWcm ⁻²	NA	93% of Toluene and 78% of Benzene	Calculated band gap: 3.11 eV for	 The toluene and benzene degradation enhanced with increase UV light intensity in a different amount of catalyst and times. The removal efficiency of benzene was less than that of toluene. Toluene & benzene removal on TiO₂/ZnO catalyst under UV light followed by first-order kinetic reactions rate. 	[141]
Benzene, Toluene, o, m and p- Xylene (BTX) (NA)	N doped TiO ₂ - Nb ₂ O ₅ composite; Method: Sol- gel	N: 0.49 at.% TiO ₂ : 50 wt.% Nb ₂ O ₅ :50 wt.%	T=25±2 °C; RH=NA; Lamp: White LED (16 W) and Black (46 W)	78.5% for TiO ₂ (UVA) 19% for TiO ₂ (Vis)	p-Xylene: 97.5% for N- TiO ₂ , 84.6% for N- TiO ₂ /Nb ₂ O ₅ (UVA) 62% for N- TiO ₂ , 56% for N- TiO ₂ /Nb ₂ O ₅ (Vis)	11O ₂ -500 C; 2.78 eV for N- TiO ₂ -500 °C; 3.28 eV for TiO ₂ -Nb ₂ O ₅ -500 °C and 3.02 eV for N- TiO ₂ -Nb ₂ O ₅ -500 °C	 The photo-oxidation reactions were followed pseudo first-order kinetics. The N-Ti calcined at 500 °C had the highest efficiency under UVA and visible light, with an apparent reaction rate constant of 0.0243 and 0.0076 min⁻¹, respectively. According to results, the heterojunction of TiO₂ and Nb₂O₅ was not effective for the photocatalytic performance. The anatase phase was detected in the TiO₂-Nb₂O₅ mixed catalysts. Results showed that all the efficiencies of N-doped catalysts were better than its undoped TiO₂ catalyst. 	[25]
Benzene, Toluene, and Xylenes (BTX) (100 ppm each)	N doped ZnO- TiO ₂ composite; Method: Sol- gel	50 wt% of ZnO and 50 wt% of TiO ₂	T=25 °C; RH=NA; Lamp: white LED, 16 W (439.8 nm);	NA	86% for Benzene, 87% for Toluene, and 86% for Xylene	3.36 eV for ZnO-	 Doped photocatalysts exhibited higher specific surface area. The presence of nitrogen led to control nucleation and growth of crystallites, as well as the formation of a well-order porous structure. The N-doped TiO₂ had visible light absorption due to the interstitial doping. The N-TiO₂-ZnO composite caused to photosensitize ZnO in visible wavelength region and reduce recombination rate of charge carriers. The highest efficiency was achieved by N-TiO₂-ZnO calcined at 500 °C. 	[105]

T: Temperature RH: Relative Humidity AFR: Air Flow Rate RT: Residence Time

NA: Not Available

