

HHS Public Access

Author manuscript Adv Exp Med Biol. Author manuscript; available in PMC 2018 March 06.

Published in final edited form as:

Adv Exp Med Biol. 2016; 940: 167–177. doi:10.1007/978-3-319-39196-0_8.

Protein-based hydrogels for tissue engineering

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Abstract

The tunable mechanical and structural properties of protein-based hydrogels make them excellent scaffolds for tissue engineering and repair. Moreover, using protein-based components provides the option to insert sequences associated with the promoting both cellular adhesion to the substrate and overall cell growth. Protein-based hydrogel components are appealing for their structural designability, specific biological functionality, and stimuli-responsiveness. Here we present highlights in the field of protein-based hydrogels for tissue engineering applications including design requirements, components, and gel types.

Structural and Mechanical Properties

To function properly and to be compatible with tissue growth, hydrogels for tissue engineering must meet a number of requirements, both physical and biological. The mechanical and structural properties of the gel - tensile strength, stiffness, elasticity, and so on - must be matched to the specific tissue type [1]. The elastic modulus (the ratio of tensile strength to tensile stress) has been shown to directly influence to cell growth and tissue development [2]. Engler *et al.* investigated the influence of the elastic modulus on cell growth and development by attaching mesenchymal stem cells (MSCs) to substrates with different elastic moduli (see Figure 1a). MSCs are derived from marrow and can differentiate into various anchorage-dependent cell-types, including neurons, myoblasts and osteoblasts. The substrates were chosen to have moduli which mimicked the brain (0.1–1 kPa), muscle (8–17 kPa), and stiff cross-linked collagen matrices (25–40 kPa) (see Figure 1b) [3]. The experiments showed that cells seeded on gels with brain-like, muscle- like and collagen matrix-like viscoelastic properties showed similar morphologies and RNA transcript profiles to neurons, myoblasts and osteoblasts respectively.

Self-Assembly

The desired mechanical and structural properties, such as a cell-type specific elastic modulus, of a protein hydrogel scaffold can be engineered by fine-tuning the self-assembly process. Obas *et al.* [4] created hydrogels with elastic moduli tunable by the alteration of

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ionic strength. As scaffolds, they used a 20 amino acid building block β -hairpin molecule (named MAX1), shown in Figure 2a, which consists of alternating valine and lysine residues connected with a tetrapeptide turn promoting sequence. The authors hypothesize that at high ionic strength, the positive charges of the lysine residues are screened and an intramolecular folding event is favored, resulting in a β -hairpin molecule with buried lysine residues and

solvent-exposed valine residues. In such a scenario, intermolecular interactions between valine residues on monomers result in aggregation into a β -sheet rich quaternary structure and ultimately hydrogel formation. MAX1 (2% by wt) solutions in the absence of NaCl did not form gels, whereas at concentrations of 20, 150 and 400 mM NaCl, gels formed with elastic moduli of ~100 Pa, ~300 Pa and ~3000 Pa respectively. Thus, increasing ionic strength not only triggers self-assembly but also results in gels with increased rigidity.

Self-assembly of α -helical peptides can also lead to hydrogel formation. Banwell *et al.* designed hydrogel scaffolds comprised of non-covalently cross-linked fibrils formed from short, α -helical peptides [5]. These hydrogelating self-assembling peptides (hSAFs) are formed by two 28-residue peptides designed to assemble into out of register, α -helical coiled-coil fibrils with complementary sticky ends. These α -helical coiled-coils are proposed to assemble longitudinally into fibrils, which then bundle to form mature fibers (Figure 3).

The hSAFs were based on previous designs from the lab based on the coiled-coil heptad repeat abcdefg [6]. The a,d,e, and g positions facilitate dimerization, while b, c, and f are surface-exposed. The b, c, and f positions were mutated to various combinations of alanine and glutamine. From these designs, hSAF_{AAA} and hSAF_{QQQ} both formed hydrogel networks. Using rat adrenal pheochromocyoma (PC12) cells, the therapeutic potential of the hSAF_{AAA} hydrogel network was tested. For these cell biology experiments, a mutation was made at position f replacing an alanine with tryptophan (hSAF_{AAA}-W), allowing for easier quantification of peptide concentration. Serendipitously, this mutation resulted in enhanced stability and gelation at room temperature. PC12 cells were then seeded on hSAF_{AAA-W} hydrogels alone and treated with nerve growth factor. The cells proliferated and showed signs of differentiation into neural cells, indicated by the visualization of neurite projections [5].

Stimuli-responsiveness

In addition to controlling the self-assembly process, the development of "smart" hydrogels that are able to respond to a stimulus can be useful in many applications, such as targeted delivery of cargo. Grove *et al.* demonstrated the formation of self-assembling hydrogels by taking advantage of an ionic interaction between a protein and peptide [7, 8]. Tetratricopeptide repeat (TPR) domains were concatenated in arrays that formed non-covalent cross-links with their corresponding peptides attached to four-arm star polyethylene glycol (PEG) molecules (Figure 4). The binding interactions forming the junctions of the network were both pH- and ionic strength-dependent, allowing for reversible gelation in response to external stimuli. When placed in solutions containing 500mM NaCl, the hydrogel dissolves, but gelation can be reconstituted through the removal of salt through dialysis [8]. This reversibility can also be demonstrated by alteration of pH [7].

Hydrogelation occurred when components were mixed in a 1:2 stoichiometric ratio (TPR arrays: PEG-peptides). The elastic modulus was measured to be \sim 270 Pa, which is well above the minimum value necessary (50 Pa –100 Pa) to support mammalian cells in suspension, demonstrating potential in applications for tissue engineering and regeneration [7].

Temperature is an alternative stimulus for controlling hydrogelation properties. Glassman *et al.* developed a thermosensitive hydrogel that is injectable at ambient temperature but exhibits enhanced stiffness and durability at higher, physiological temperatures [9]. This heat-sensitive reinforcement is derived from two separate networks that contribute to the overall properties of the hydrogel. Junctions formed by coiled-coil associations constitute the shear-thinning hydrogel, while the endblocks are comprised of a thermoresponsive polymer called poly(*N*-isopropylacrylamide) (PNIPAM). The engineered coiled-coil segment was flanked by PNIPAM endblocks (Figure 5). The coiled-coils interact to form a shear-thinning hydrogel, which is reinforced at higher temperatures due to association of the PNIPAM heat-sensitive domains. A shear-thinning hydrogel that transitions to a more rigid network at physiological temperatures offers practical use for tissue repair via injection.

Pore Size

Gel scaffold pore size has been hypothesized to both influence nutrient transport to implanted cells and to guide growth in three dimensions. Zeltinger *et al.* [10] investigated the effects of pore size on the attachment, proliferation, extracellular matrix formation, and cellular metabolism of canine skin and vascular cell types. They generated hydrogel scaffolds with four different pore sizes (<38 μ m, 38–63 μ m, 63–106 μ m, and 106–150 μ m), determined by scanning electron microscopy (SEM). Each cell type required a particular pore size for optimal growth; microvascular epithelial cells required a pore size <38 μ m to form a thin endothelial lining, vascular smooth muscle cells preferred scaffolds >63 μ m to form uniform tissue, whereas the growth of dermal fibroblasts was independent of pore size. Other experiments have also shown that optimum pore size is cell-type specific and can range from 5 μ m for neovascularization [6] to greater than 500 μ m for fibrovascular tissue [7].

Controlling the pore size in protein-based gels requires the tuning of both component concentration [9,10] and the physical conditions of gelation such as temperature [11] or ionic strength [10]. In some instances, finer control of pore size can be accomplished through the use of a rapid freeze-drying technique [11], which is well-illustrated by the work of Mandal *et al.* [9] in the fabrication of silk fibroin 3D scaffolds with different pore sizes. The researchers created porous protein scaffolds by pouring regenerated silk solutions (2–6wt %) in Teflon molds. The solutions were then frozen for 24 hours at -20, -80 or -196 °C by keeping the bottom of the mold in contact with the freezer. Uniform directional cooling was accomplished by sealing all sides, except the bottom of the mold results in the uniform and directional formation of pores within the scaffold by ice crystal formation. Pore size decreased directly as a function of temperature and inversely as a function of silk fibroin concentration (see Table 1).

Another critical component of hydrogel-based tissue engineering is the ability of the system to mimic the function of the extracellular matrix (ECM) by presenting factors necessary to direct tissue formation and regeneration [16]. This can be done through the incorporation of growth factors, which are signaling molecules that can direct cell proliferation, differentiation, healing, and migration [17]. By controlling pore size, Lindsey *et al.* designed a hydrogel delivery system for nerve growth factor (NGF) and brain-derived neurotrophic factor (BDNF) [18] using the self-assembling peptide-based hydrogel MAX8 [19]. MAX8 peptide hydrogels are derived from the previously mentioned MAX1 peptides and have unique shear-thinning properties that are attractive for cargo delivery via injection [29]. Their studies showed that encapsulation of growth factors did not affect the physical properties of MAX8, and the rate of growth factor release was dependent on the growth factor concentration within the gel. Additionally, the pore size of the hydrogel decreased inversely with the concentration of MAX8 peptides, allowing tunable encapsulation and release of molecular cargo.

Degradation

The controlled degradation of hydrogel scaffolds is an important physical parameter in tissue engineering. Hydrogels used for implantation must degrade, at a rate that is compatible with the desired function, into non-toxic products that can be easily cleared from the body, leaving healthy tissue post degradation [18]. Depending on the nature of the gel components, degradation can be primarily the result of enzymatic cleavage, hydrolysis or dissolution. Hydrogels formed by covalent cross-links between components must be degraded by enzymatic cleavage or hydrolysis, whereas gels formed by non-covalent physical interactions between components may be degraded by dissolution [19].

Balakrishnan *et al.* developed an injectable hydrogel with cross-links produced by a Schiff's base reaction between the lysine or hydroxylysine side chains of gelatin and the aldehyde on the chemical alginate dealdehyde (ADA) catalyzed by $Na_2B_4O_7$ (borax) (see Figure 7) [20]. To mix the components with the potential to form the gel *in situ*, researchers used a double-syringe fibrin glue applicator with one syringe filled with the gelatin solution and the other with oxidized ADA plus borax. Both solutions are pushed into a hypodermic needle where the polymers mix, gel, and cross-link within seconds. The susceptibility of the Schiff's linkage to hydrolysis, and the degradability of ADA and gelatin by the body make the scaffold fully biodegradable over time.

Hydrogels designed to undergo degradation by enzymatic cleavage are well-illustrated by the work of Galler *et al.* with a multidomain peptide (MDP) hydrogel [21]. MDPs are amphiphilic peptide systems with amino acids arranged in a block motif that self-assemble into nanofibers. The core interactions of supramolecular assembly are driven by the "B" block, which consists of alternating hydrophobic (leucine, L) and hydrophilic (serine, S) residues (see Table 2). In aqueous environments, the side chains present on opposite sides of the peptide backbone and two peptides can come together at the hydrophobic face to create a dimeric "sandwich" stabilized by the packing of the leucine residues (Figure 8) [22]. The dimers are proposed to interact through anti-parallel β -sheet hydrogen bonding oriented down the fiber axis (Figure 8). The flanking "A" block has charged amino acids, lysine (K),

that create water solubility as well as an electrostatic repulsion that works against fiber formation when unshielded, thus can be manipulated to control fiber assembly and length [23]. The addition of multivalent ions can be used to screen these charges and allows fiber elongation beyond dimerization and eventual gel formation.

To encode stimuli-responsiveness into this design, the researchers created a peptide arranged in an ABA block motif containing the matrix metalloproteinase 2 (MMP-2) leucinearginine-glycine consensus cleavage motif, see peptide MDP3 in Table 2 [24]. Gelation of the peptide was induced by the addition of phosphate buffered saline. Gels containing the MMP-2 cleavage sequence, (i.e. made using peptide MDP3), degraded upon treatment with MMP-2 whereas gels without the cleavage sequence, MDPs 1 and 2, did not. MDP2 included the well-known amino acid cell adhesion motif RGD to increase the hydrogel compatibility for cell culture. RGD sequences have been shown increase cell spreading, proliferation and migration within hydrogel scaffolds [25].

Another example of enzymatically degradable hydrogels is demonstrated by the work of Sun *et al.* using SpyTag/SpyCatcher technology [26]. SpyTag and SpyCatcher are peptide and protein components, respectively, that originated from splitting a fibronectin-binding domain of a bacterial adhesin and engineering it into separate parts [27]. Upon mixing, SpyCatcher and SpyTag spontaneously form an isopeptide linkage between the sidechains of a lysine on SpyCatcher and an aspartate on SpyTag. This has been shown by Zhang *et al.* to be a powerful tool for covalent attachment and the creation of interesting protein topologies [28].

By encoding SpyTag and SpyCatcher into specific parts of ELP sequences, Sun *et al.* designed components intended to form covalent linkages leading to hydrogelation. The first construct, called AAA, was comprised of three SpyTag sequences separating by two ELP sequences (Figure 9). The second construct, referred to as BB, consisted of an ELP sequence flanked by two SpyCatchers (Figure 9). The BB construct also contained an internal integrin-binding RGD sequence and an MMP-1 cleavage site embedded in the ELP region, with the goals of cellular adhesion and potential of enzymatic degradation [26]. To demonstrate potential in tissue engineering applications, Sun *et al.* showed that the hydrogels could encapsulate mouse 3T3 fibroblasts. After growth in the matrix for 24 hours, the majority of cells were still alive and displayed "elongated morphologies", suggesting that cellular adhesion and matrix remodeling had occurred [26]. This design illustrates one major advantage of protein-based hydrogels, which is that functionality can be encoded into the sequence in a straightforward way.

Biocompatibility

Hydrogel biocompatibility and low immunogenicity are essential for *in vivo* applications. Researchers must either choose components with known biocompatibility or engineer components to be more compatible. For protein-based hydrogels, site-specific mutagenesis and/or truncation can be used by researchers to remove immunogenic epitopes, while maintaining the maintenance gelation properties [29]. The ease of manipulation of proteinbased materials in this fashion to fine-tune their properties is a unique and attractive feature. Some proteins, such as elastin-like polymers (ELPs), naturally exhibit low immunogenicity

[30] and can confer this biocompatibility onto tissue engineering scaffolds. Testera *et al.* used ELPs as protein-based polymer building blocks for a tissue engineering hydrogel scaffold [31]. Using click chemistry, ELPs were decorated at lysine residues with either alkyne or azide groups to create two gel components (see Figure 10). Mixing of the two components resulted in cross-linking between azide and alkyne groups, forming hydrogels that were able to maintain long-term viability with respect to human primary cell types in both surface and 3D cultures, thus showing the importance of hydrogel scaffold biodegradability to viability.

Conclusion

Protein-based hydrogels are excellent candidates tissue engineering and repair scaffolds because of the ease of designability. Here, we have shown that researchers have been able to use sequence-to-structure information to take advantage of the self-assembly process as well as create hydrogel scaffolds with tunable elastic moduli, stimuli-responsiveness, varying pore sizes and increased biocompatibility. As the field of protein engineering continues to develop and create new tools, protein-based hydrogel systems will continue to develop as an innovative tool for tissue engineering.

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Fig. 1.

(a) Schematic illustration of cells grown on elastic substrates with varying elastic moduli. Collagen-I was used as the adhesion ligand (red triangles) to promote cell adhesion to gels through via interactions with cellular integrins. (b) Scale showing a range of elastic moduli for different tissue types



Fig. 2.

(a) Proposed hairpin structure of the 20 residue long MAX1 peptide. (b) Cartoon representation of intramolecular folding followed by self-assembly in response to increased ionic strength



Fig. 3.

(a) In previous SAF designs, the solvent exposed residues b and c contained specific charged residues, which led to helix alignment and thick fiber formation. (b) In hSAF designs, the b, f, and c positions were changed to combinations of glutamine and alanine with the goal of achieving weaker interactions, leading to small, flexible bundles of fibers

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Fig. 4.

A cartoon showing the reversible formation of a TPR hydrogel network. TPR arrays are shown as blue rods with alternating binding modules (navy) and spacer modules (cyan). Corresponding peptides (red) are displayed on the ends of 4-armed PEG star molecules



Fig. 5.

(a) An illustration of the shear-thinning hydrogel (left) which is reinforced to become a stiffer, more rigid network at higher temperatures. (b) A cartoon of the design of the dual-system hydrogel components





Schematic representation of directional ice crystal formation within silk fibroin hydrogel scaffolds. Cylinder representation shows upward ice crystal formation over time and circular cross-sections show inward ice crystal formation over time



Alginate dialdehyde cross-linked gelatin hydrogel

Fig. 7. ADA cross-linking with gelatin in the presence of borax $(Na_2B_4O_7)$



Fig. 8. Cartoon illustration of MDP fiber formation



Fig. 9.

A cartoon illustrating the placement of SpyTag and SpyCatcher on the two constructs labeled AAA and BB. Covalent linkages via isopeptide bonding form the hydrogel network. Gelation occurred within 5 min of mixing AAA and BB at 10 wt % in a 2:3 molar ratio at room temperature





Table. 1

Pore sizes obtained under different silk fibroin concentrations and freezing conditions

Scaffold temperature (°C)	Silk fibroin concentration (wt %)			
		2	4	6
	Pore size range (µm)			
-20		200-250	100-150	75–100
-80		100-150	75–100	50-75
-196		80-100	60-80	50-75

Table. 2

Names and sequences of peptides studied

MDP#	Sequence
1	KK-SLSLSLSLSLSLSLMKK
2	KK-SLSLSLSLSLSLSLM-KKGRGDS
3	kk-slsl SLRGslslsl-kk